

# Recent advances of iron-based superconducting wires and tapes

Yanwei Ma

Institute of Electrical Engineering,  
Chinese Academy of Sciences,  
Beijing, China



# Outline

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- 1 Background on Fe-based superconductor
- 2 Fabrication techniques to achieve high- $J_c$   
in 122 wires and tapes
- 3 Recent results about practical properties  
of 122 tapes
- 4 Conclusions

# So far, over 1000 superconductors have been discovered

## Three key properties for applications

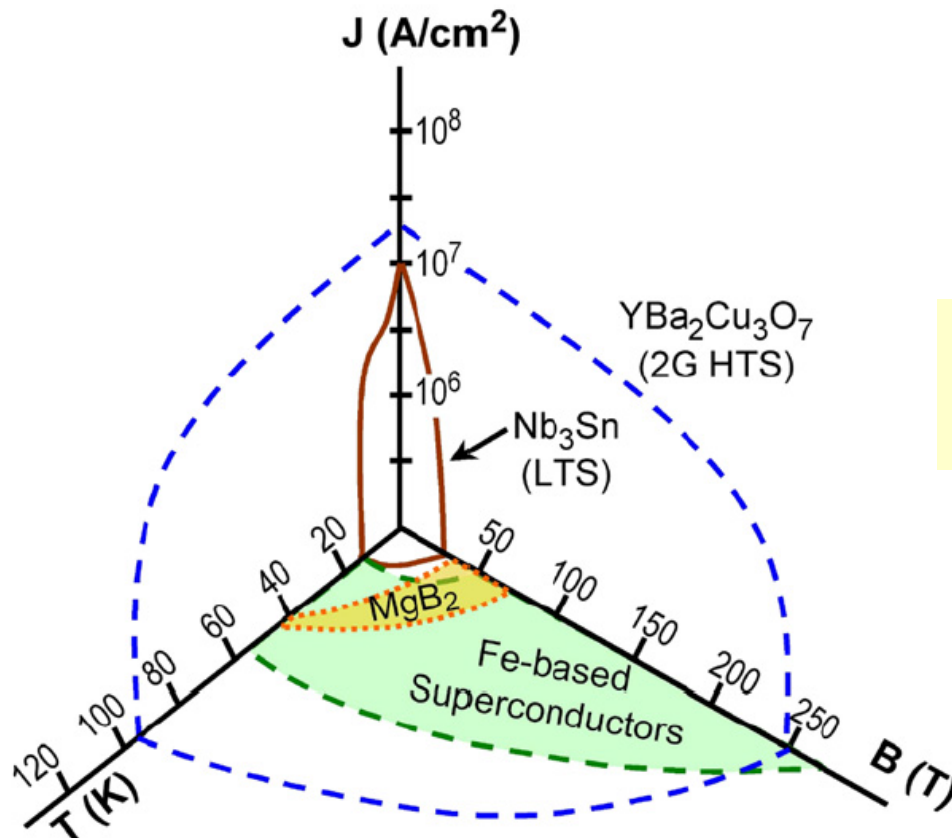
- ◆ Transition temperature,  $T_c$
- ◆ Upper critical field,  $H_{c2}$
- ◆ Critical current density,  $J_c$

For applications, besides high  $T_c$ , large  $J_c$  and high  $H_{c2}$  are required.



**Practical superconductor**

**LTS, cuprate HTS,  $MgB_2$ , Fe-base**





# Practical Wires & Tapes

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- **Commercial production:**
  - Niobium alloys (NbTi, Nb<sub>3</sub>Sn etc)
  - **Bi2223, Bi2212** / silver tape - 1<sup>st</sup> Generation HTS
  - **MgB<sub>2</sub>**
- **Pre-commercial:**
  - **YBCO** 2<sup>nd</sup> Generation HTS “coated conductor”
- **Laboratory:**
  - **Fe-based superconducting wires**

# Iron-Based Superconductors (IBS)

*J. Am. Chem. Soc.*, **130** (11), 3296 -3297, 2008. 10.1021/ja800073m

Web Release Date: February 23, 2008

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## Iron-Based Layered Superconductor $\text{La}[\text{O}_{1-x}\text{F}_x]\text{FeAs}$ ( $x = 0.05\text{-}0.12$ ) with $T_c = 26\text{ K}$

Yoichi Kamihara,<sup>†</sup> Takumi Watanabe,<sup>‡</sup> Masahiro Hirano,<sup>§</sup> and Hideo Hosono<sup>†§</sup>

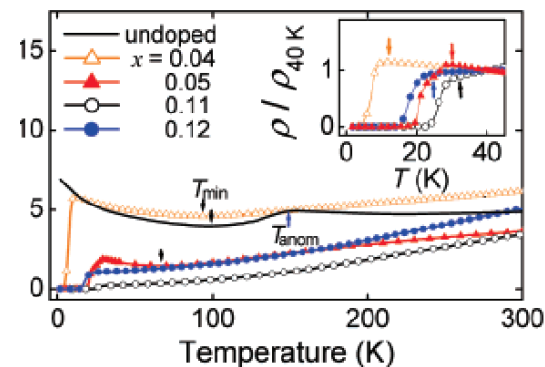
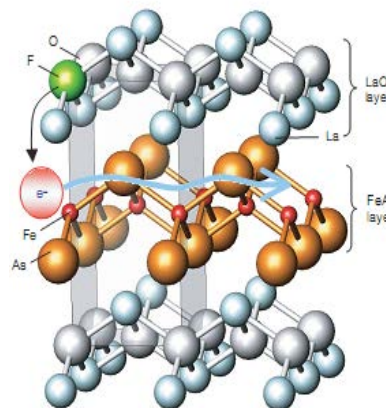
*ERATO-SORST, JST, Frontier Research Center, Tokyo Institute of Technology, Mail Box S2-13, Materials and Structures Laboratory, Tokyo Institute of Technology, Mail Box R3-1, and Frontier Research Center, Tokyo Institute of Technology, Mail Box S2-13, 4259 Nagatsuta, Midori-ku, Yokohama 226-8503, Japan*

hosono@msl.titech.ac.jp

Received January 9, 2008

### Abstract:

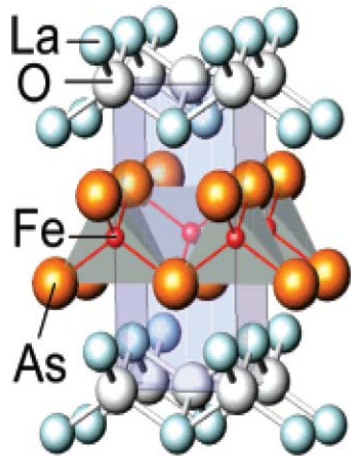
We report that a layered iron-based compound  $\text{LaOFeAs}$  undergoes superconducting transition under doping with  $\text{F}^-$  ions at the  $\text{O}^{2-}$  site. The transition temperature ( $T_c$ ) exhibits a trapezoid shape dependence on the  $\text{F}^-$  content, with the highest  $T_c$  of  $\sim 26\text{ K}$  at  $\sim 11\text{ atom \%}$ .



# Main known IBS families

Among them, the three phases most relevant for wire applications are 1111, 122, and 11 types with a  $T_c$  of 55, 38 and 8 K, respectively.

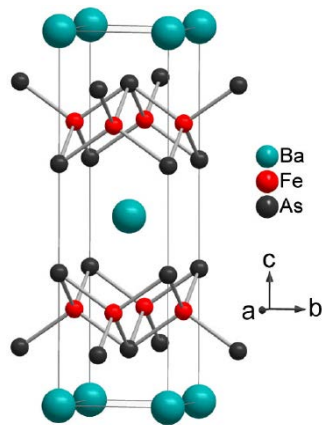
## 1111 Phase LnOFeAs



$T_c \sim 55$  K

Z. A. Ren et al., *Chin. Phys. Lett.* 25, 2215 (2008)

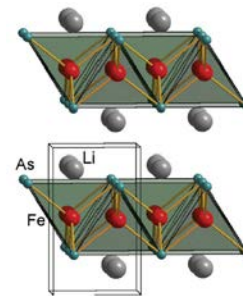
## 122 phase AFe<sub>2</sub>As<sub>2</sub> (A=Ba, Sr, Ca)



$T_c \sim 38$  K

M. Rotter, et al., *Phys. Rev. Lett.* 101, 107006 (2008)

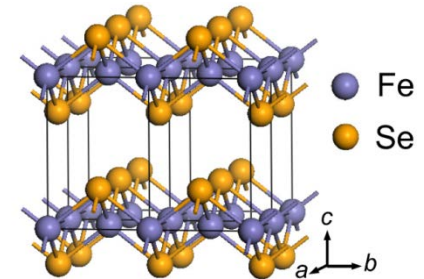
## 111 phase LiFeAs



$T_c \sim 18$  K

X. C. Wang, et al., *Solid State Commun.* 148, 538 (2008).

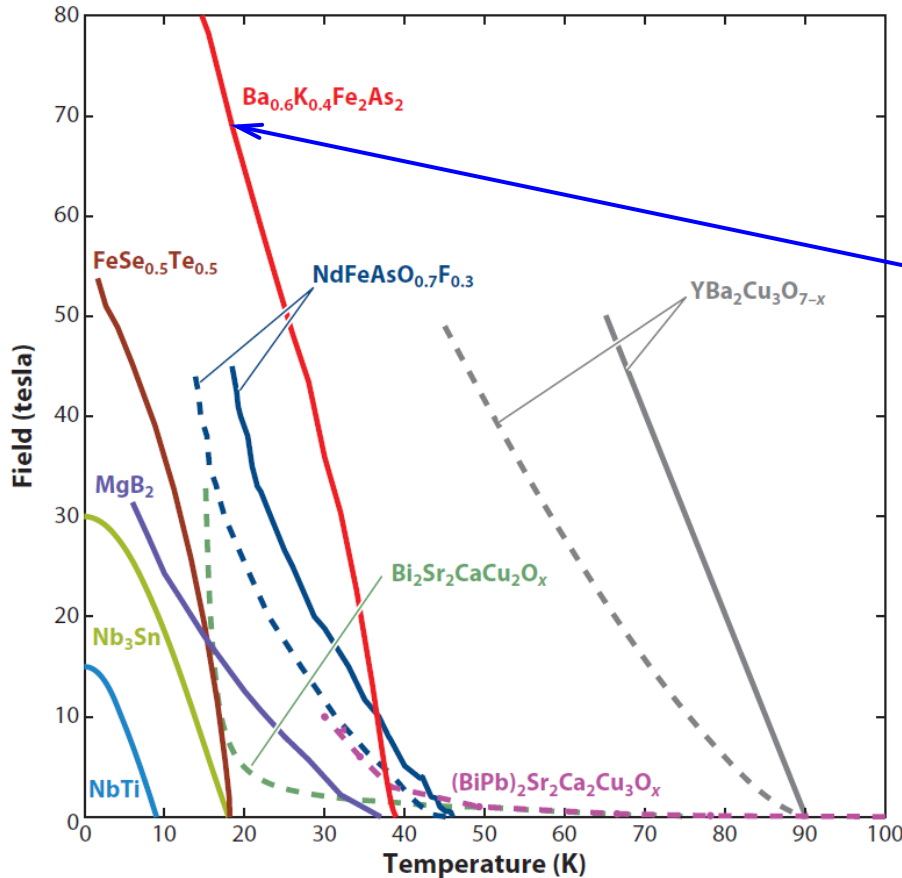
## 11 phase FeSe



$T_c \sim 8$  K

F. C. Hsu, et al., *Proc. Natl. Acad. Sci. U.S.A.* 105, 14262 (2008).

# The extremely high $H_{c2}$ in IBS



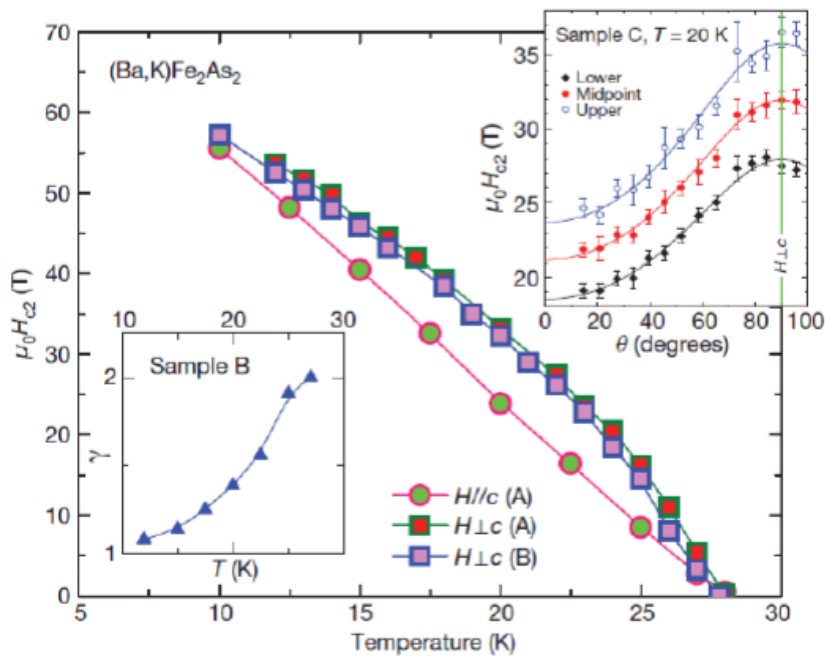
At 20 K, the  $H_{c2}$  can be  $>70$  T where IBS outperform both  $\text{MgB}_2$  and Bi-2223.

- Interesting FBS have  $T_c$ : 38-55 K  $\gg$  Nb-Ti and  $\text{Nb}_3\text{Sn}$
- Operation at 4K  $>20$ T or 10-30 K at  $>10$  T would be very valuable

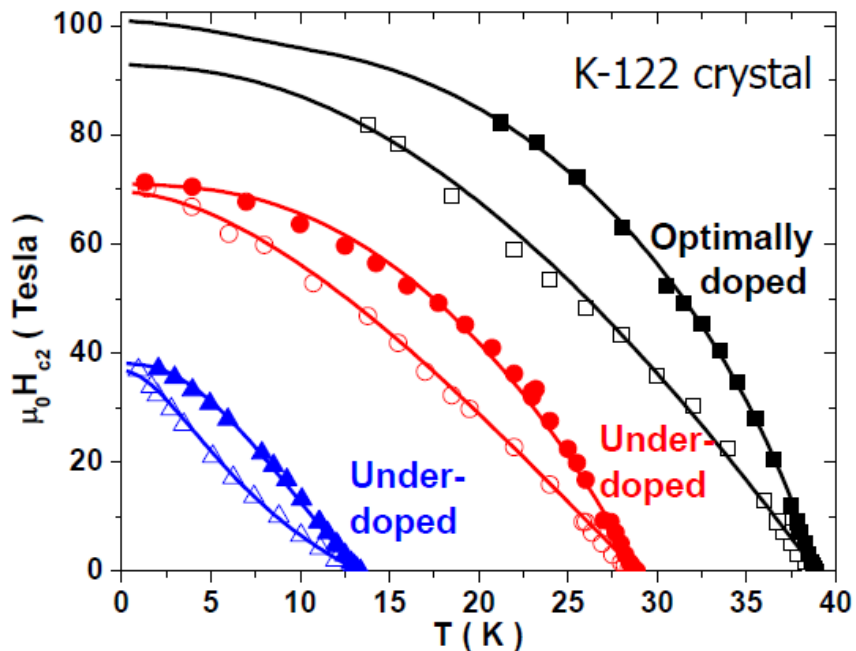
Gurevich, *Nature Mater.* 10 (2011) 255

The extremely high  $H_{c2}$  in IBS shows a great potential for applications in high field magnets, e.g.,  $H > 20$  T, which cannot be achieved via LTS and  $\text{MgB}_2$ .

# 122 IBS - small anisotropy $\gamma$



Yuan et al. Nature 457, 565 (2009)

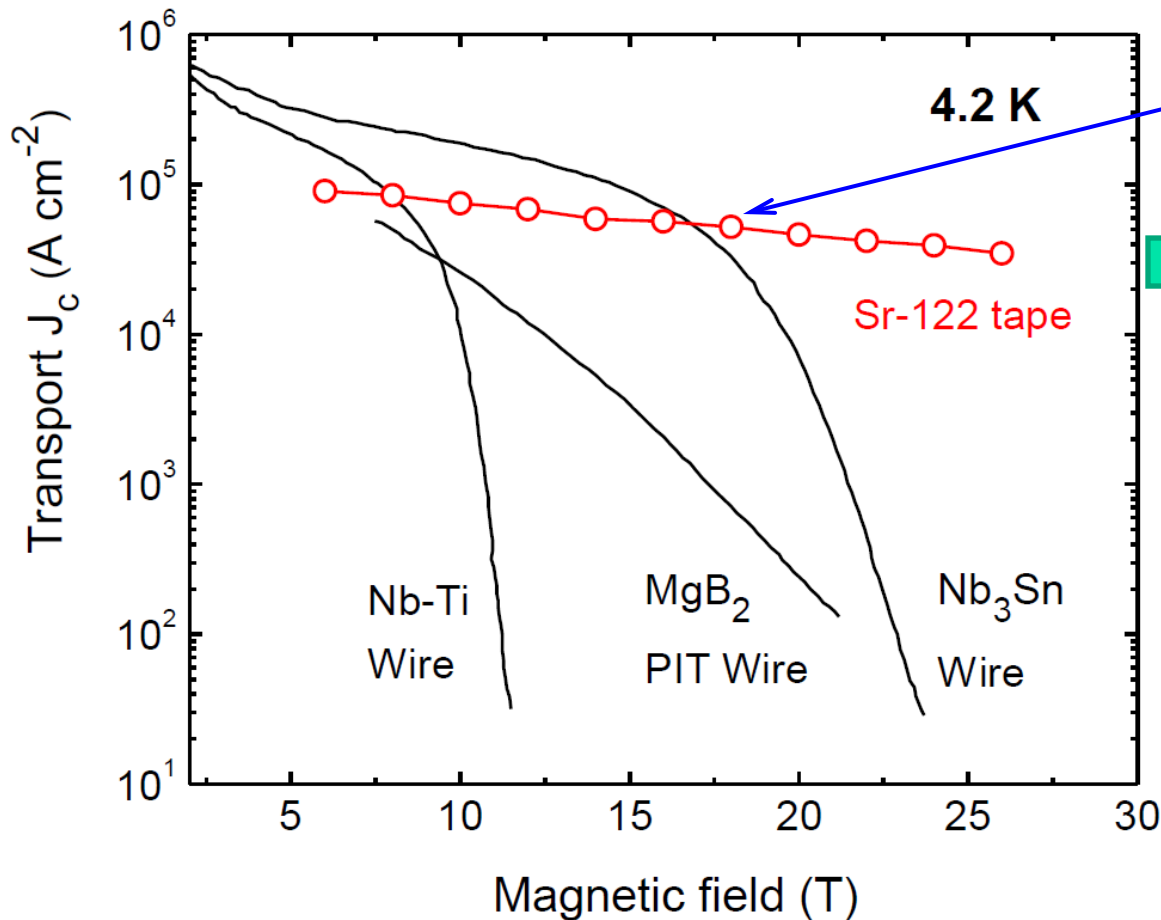


Tarantini et al. PRB 86, 214504 (2012)

- ➡  $\gamma \sim 1.1$  for K-122, nearly isotropic
- ➡  $\gamma$  is almost 1, clearly, vortices are much more rigid than in any cuprate-much easier to prevent depinning of any GB segment



# The $J_c$ of IBS wires: Very weak field dependence in high field region

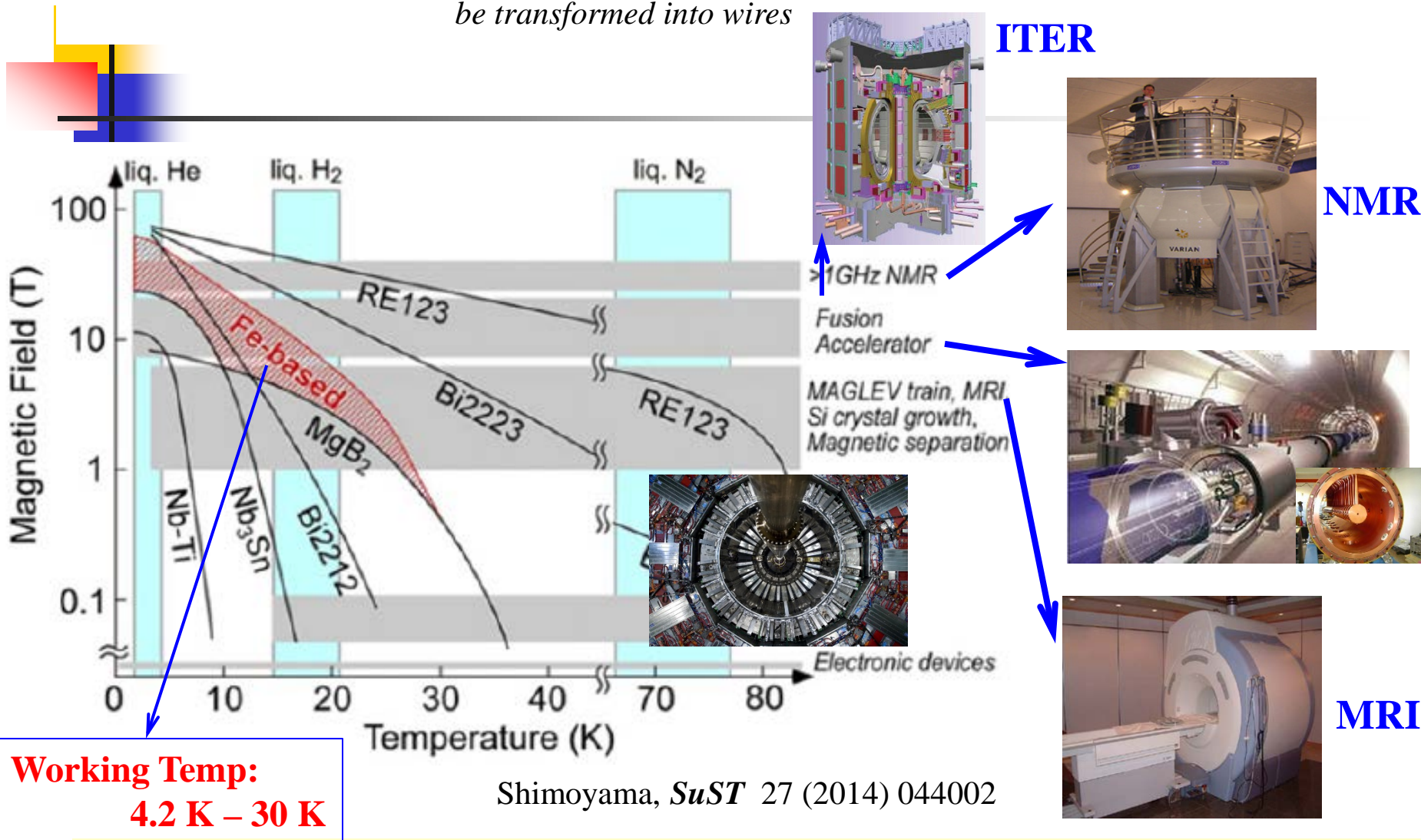


**122 IBS wire:  
Large  $J_c$ , at  $H > 20T$**

**$J_c$  shows very weak field  
dependence in high fields**

# IBS potential for high-field applications

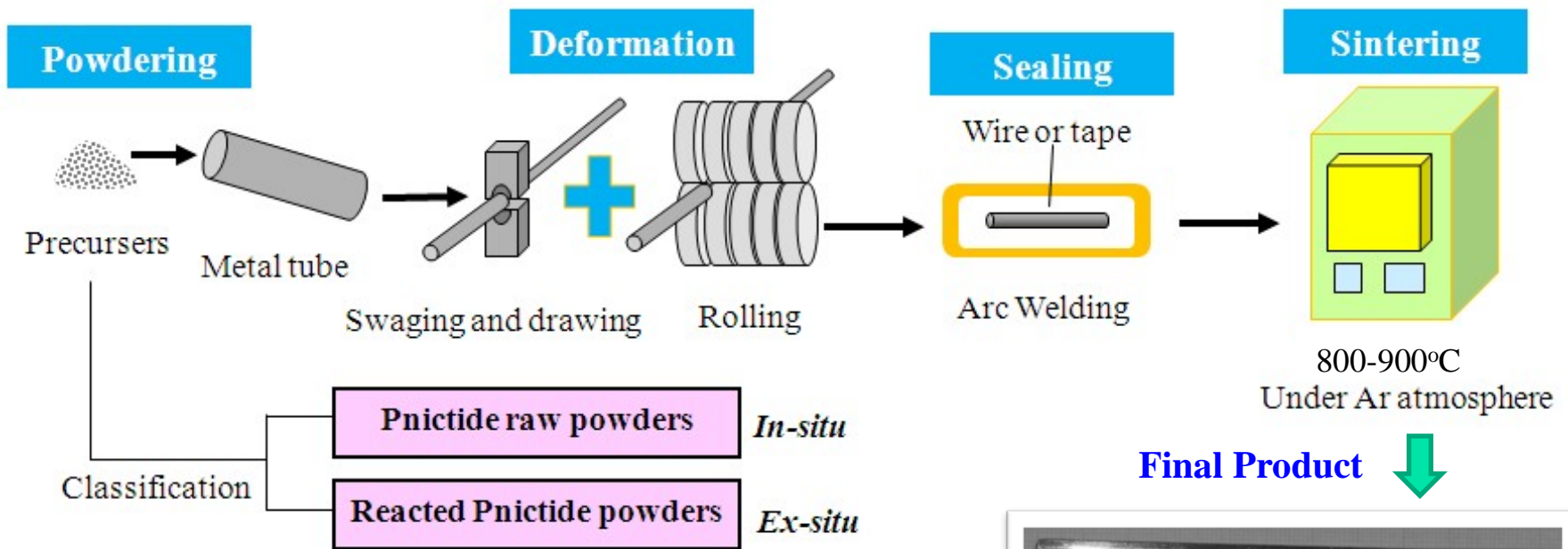
To apply superconducting materials to technologies related to magnets, they must be transformed into wires



*Development of high-performance wire conductors is essential*

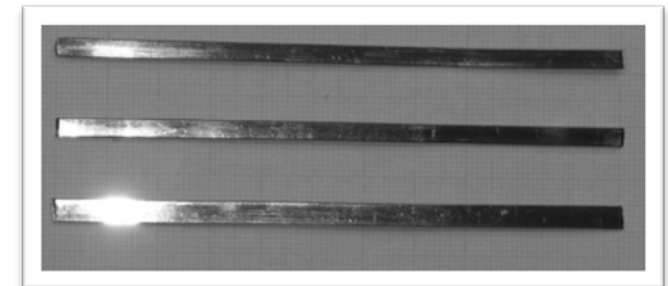
# Fabrication process for $\text{Sr}(\text{Ba})_{1-x}\text{K}_x\text{Fe}_2\text{As}_2$ wires (*Powder-in-tube method*)

— Simple and scalable process, low cost



## 122 PIT wires:

1. The single phase can easily be obtained.
2. The sintering temperature is low.



Iron-based superconducting tapes

# In April 2008, the first pnictide wire was fabricated by the powder-in-tube method

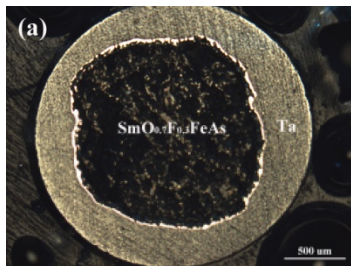
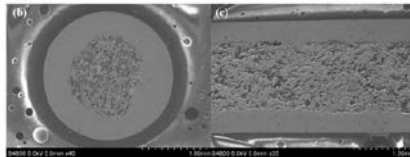
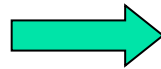
The early wires



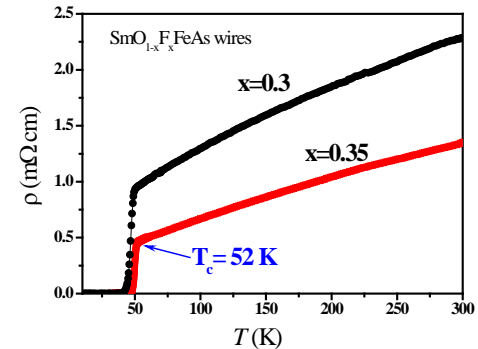
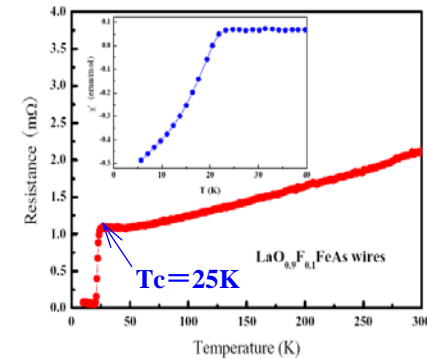
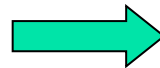
- ◆ much low critical current density  $J_c$ !
- ◆ due to thick reaction layer, many impurities, and cracks.



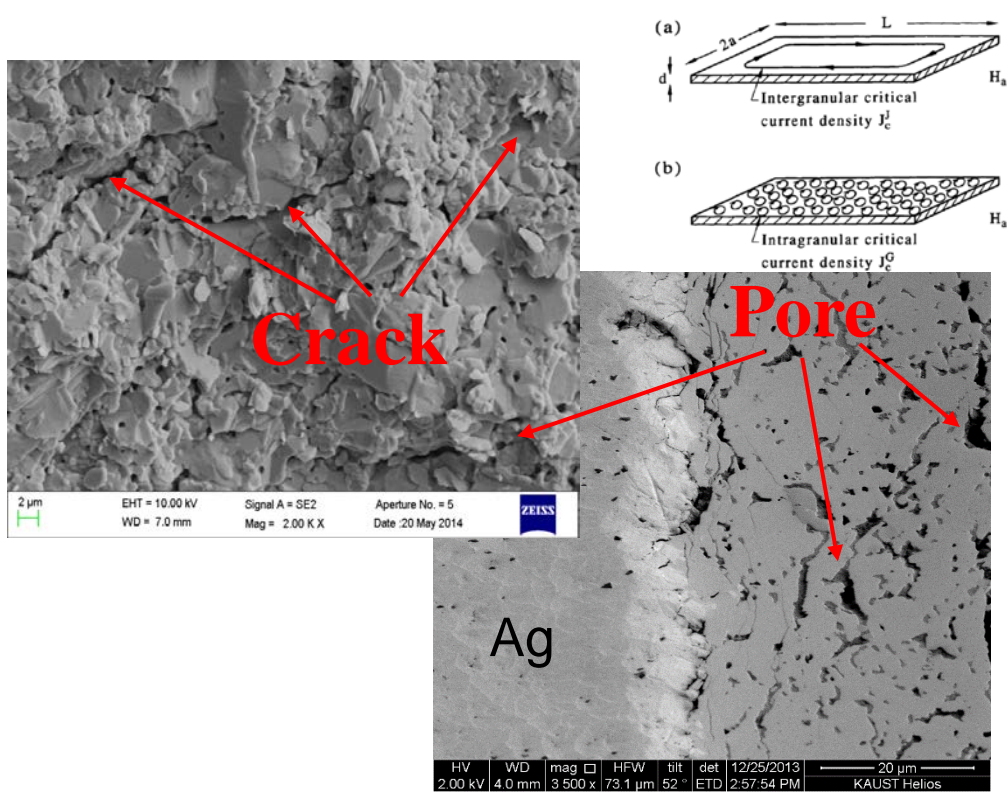
LaOFeAs



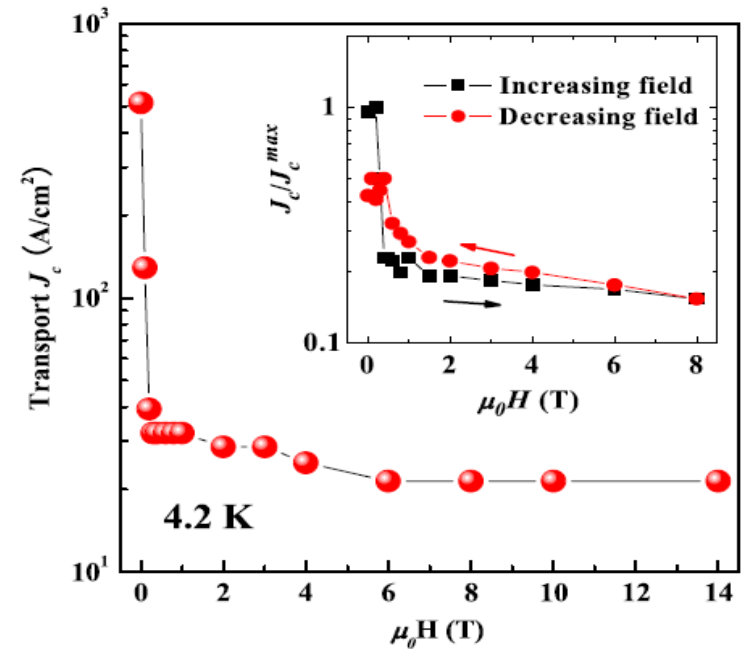
SmOFeAs



# Key problems for PIT wires: **Low density and weak link**



Good connectivity is desirable!



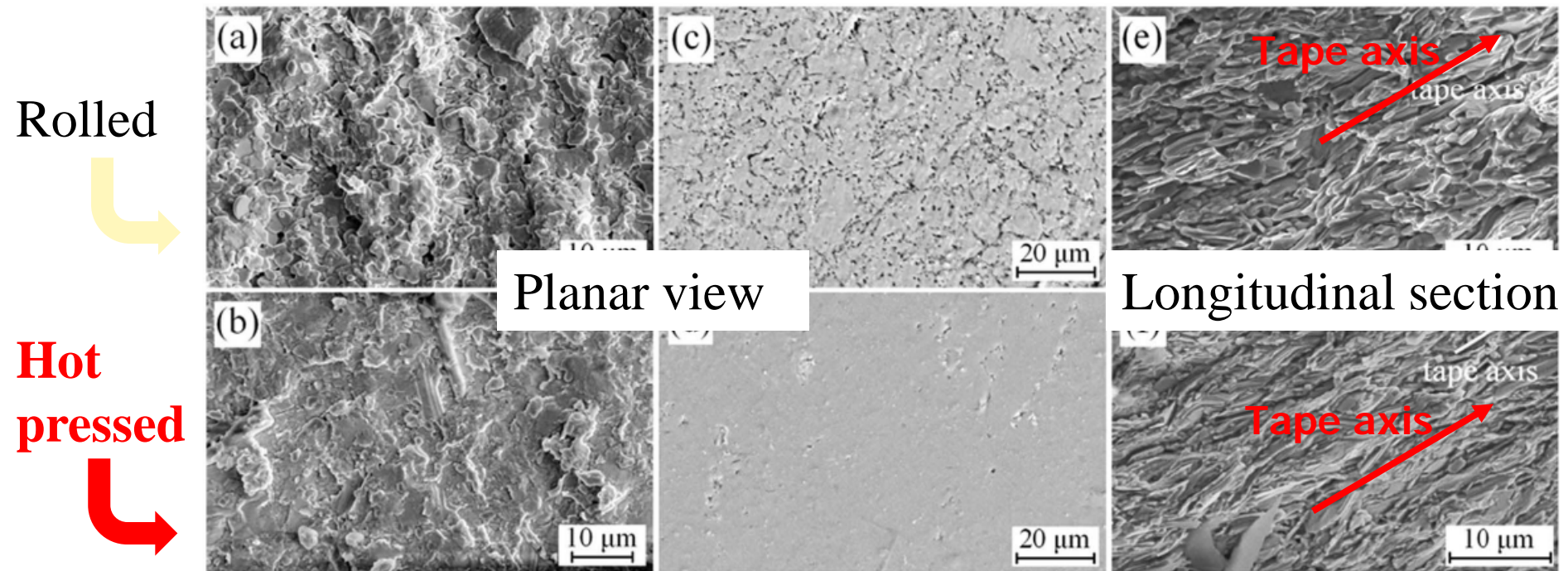
**Low density: cracks and porosity**

**Hysteresis in transport  $J_c$ : signature of weak links**

- ➡ **Residual cracks and porosity** always lead to poor grain connection, so suppress  $J_c$  in polycrystalline wires!
- ➡ A hysteretic phenomenon observed for transport  $J_c$  in an increasing and a decreasing field indicated a **weak-linked behavior**, similar to that of the cuprates.

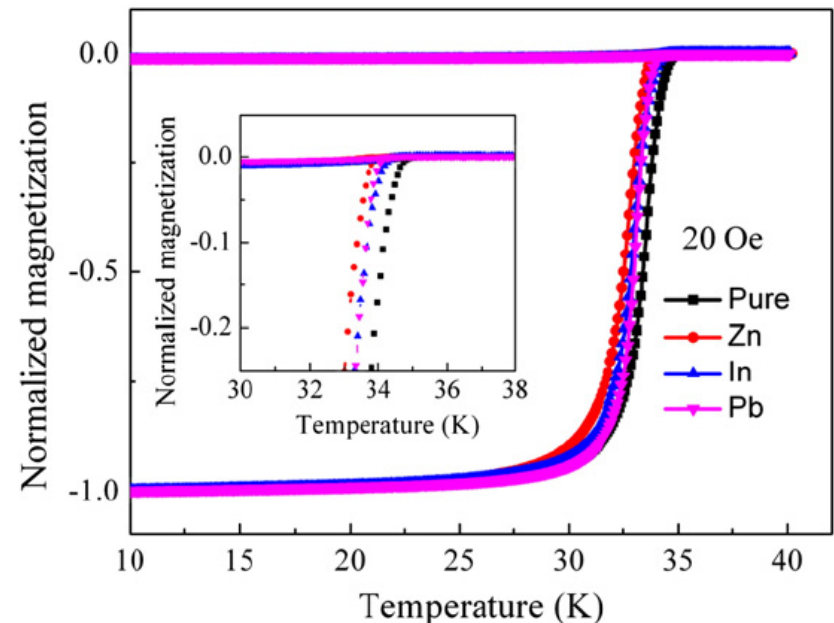
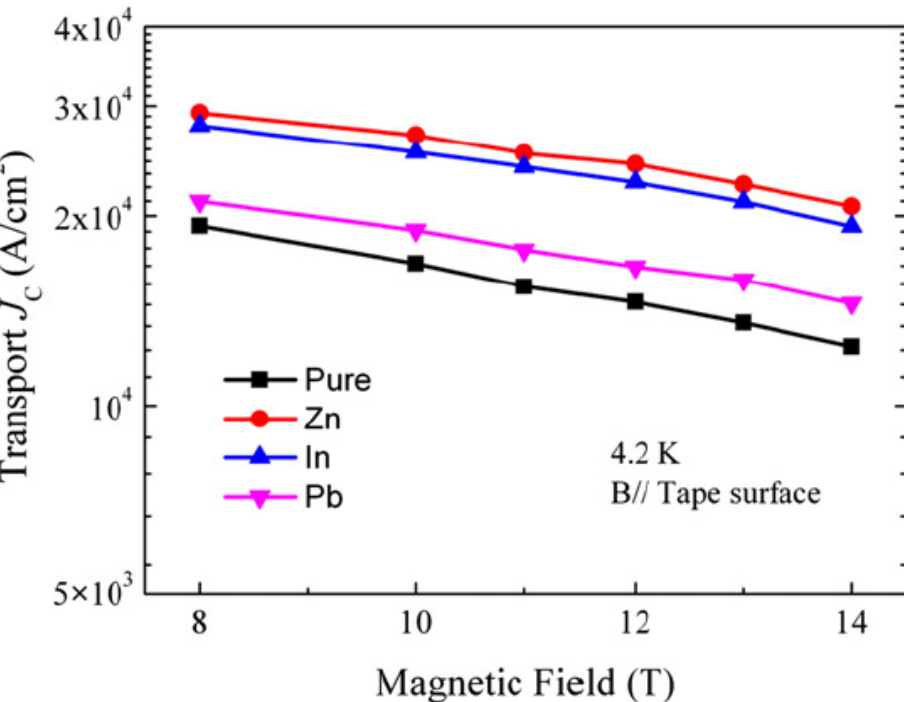
# Solutions

- ✓ **Add Sn** to improve the grain connectivity. (APL2011)
- ✓ **Large reduction rolling** to increase texture. (*Physica C* 2011, *APL* 2011)
- ✓ **Hot pressing** is effective to enhance the superconducting core density as well as grain alignment. (*APL*2014)



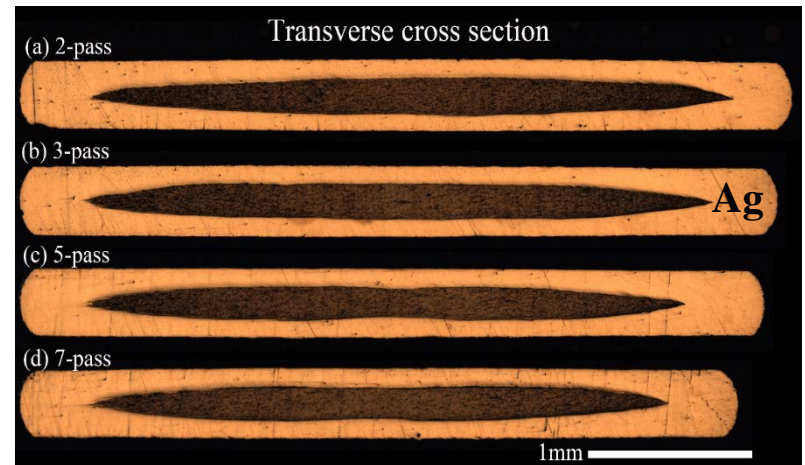
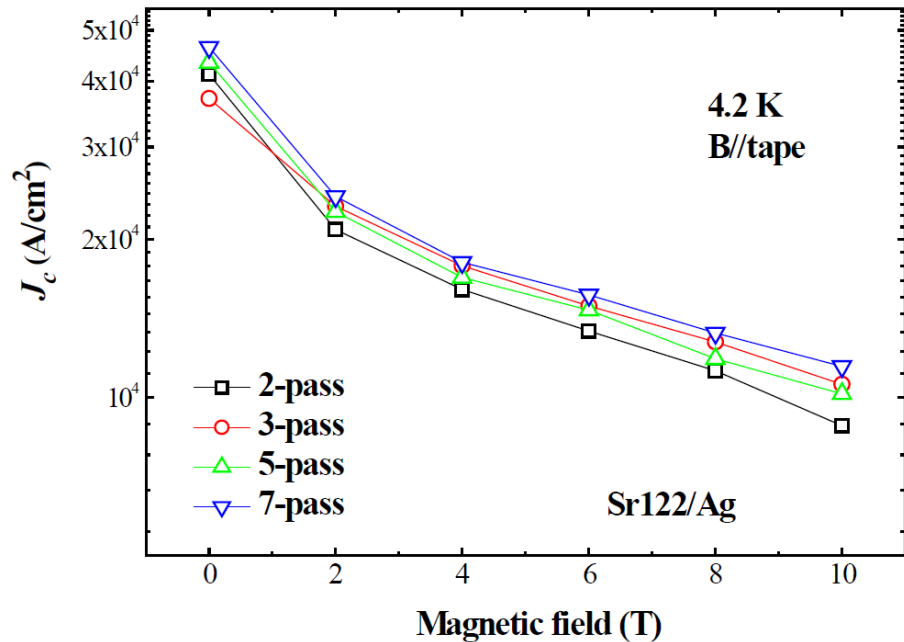
# Zn and In additions are effective to enhance the $J_c$ -B of 122/Ag tapes

Chemical addition has been confirmed as a simple and readily scalable technique for enhancing  $J_c$ .

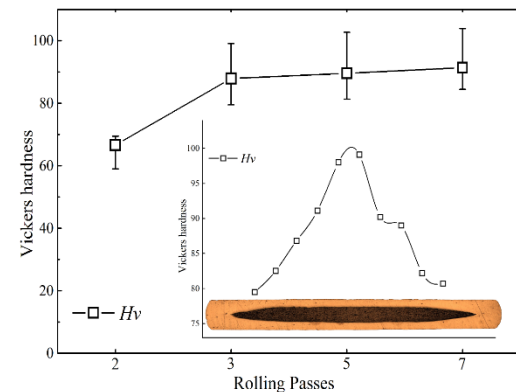


- ◆ The additions do not significantly affect the temperature transition  $T_c$ , and the  $T_c$  decreased only 0.4 K.
- ◆ the  $J_c$  enhancement in In or Zn-added samples may be attributed to the improved phase uniformity as well as the good grain connectivity

# Optimized rolling process for 122/Ag tapes: 3-pass deformation is best

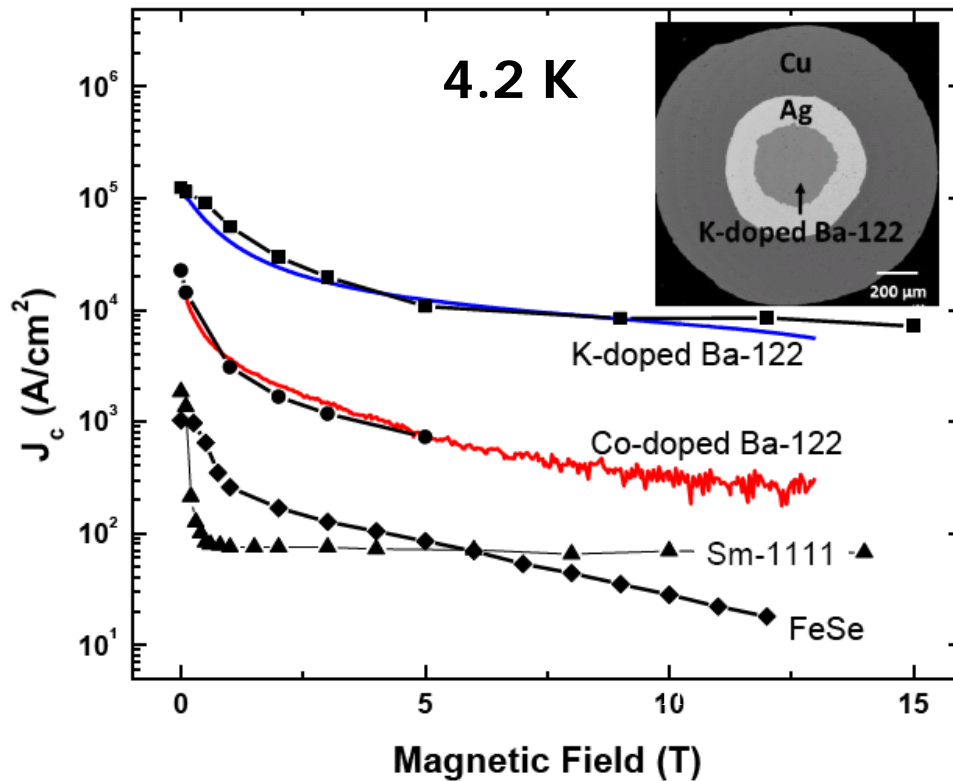


- ➔ The width of the tapes and the area of superconducting cores increase with decreasing the rolling pass, but the transport  $J_c$  seems close.
- ➔ We can fabricate tapes with 3 rolling passes to get the uniform and high- $J_c$  122 tapes.





## New synthesis method (HIP) increased $J_c$ in Ba-122 round wire



Hot isostatic press (HIP)  
under 192 MPa of pressure at  
600 ° C for 20 hours



Nearly 100% dense core



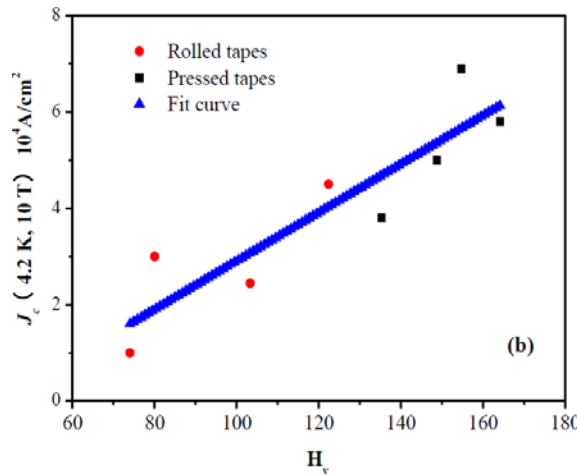
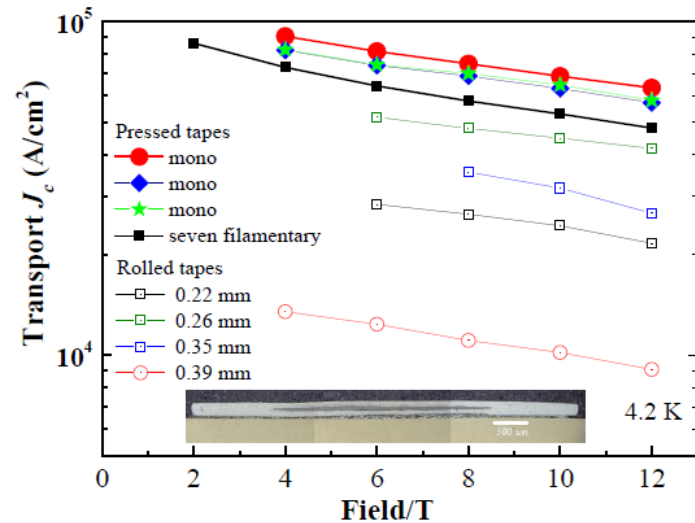
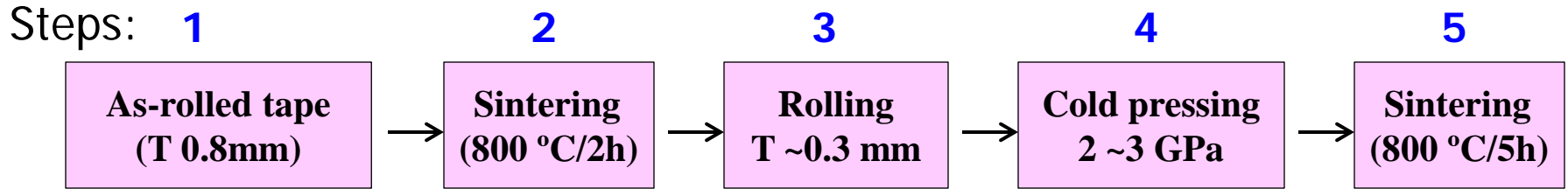
4.2 K

0 T,  $J_c > 10^5$  A/cm<sup>2</sup>

10 T,  $J_c = \sim 10000$  A/cm<sup>2</sup>

## Thin tapes by combined the rolling, cold pressing and sintering process-- Denser core yields higher $J_c$

--Ag-sheathed Ba122 tapes



The higher the core density, the higher the  $J_c$



4.2 K, 10T:

$$J_c = \sim 86000 \text{ A/cm}^2$$

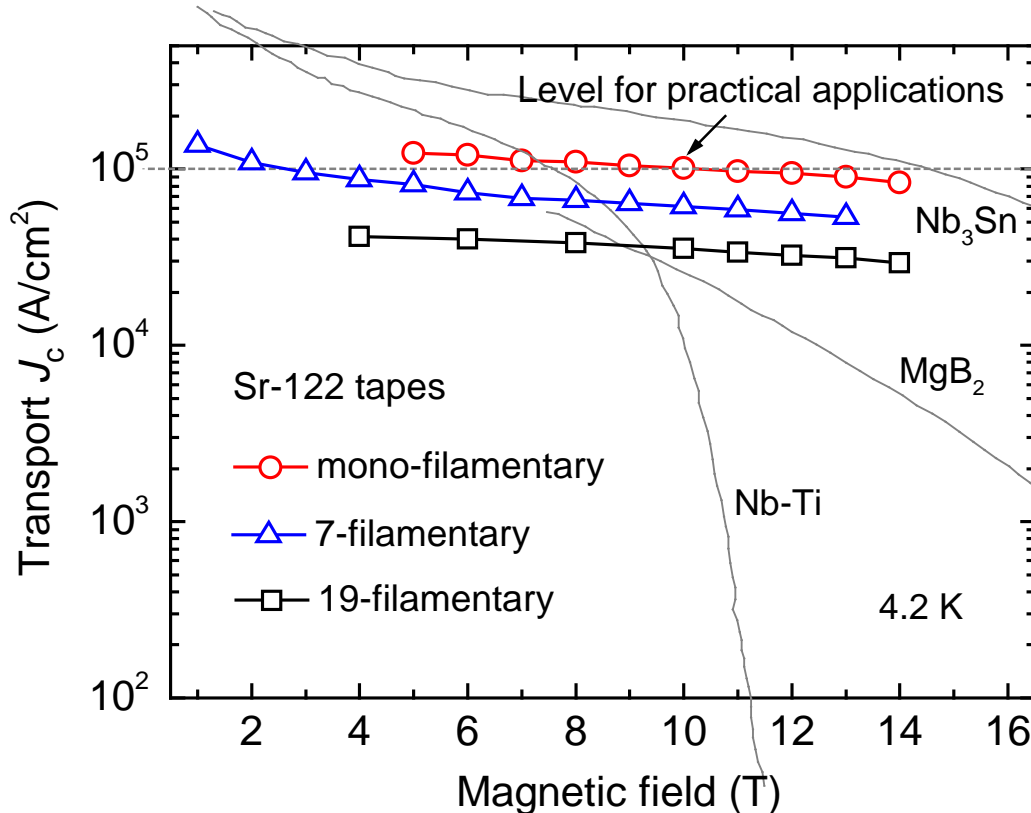
## Breakthrough work

**Very High transport  $J_c$  were achieved in 122/Ag tapes:  
 $J_c > 10^5$  A/cm<sup>2</sup> (4.2 K, 10 T) - by hot pressing**



**First to reach practical level  $J_c$ !**

The threshold for practical application:  
 $J_c = 10^5$  A/cm<sup>2</sup>@10 T



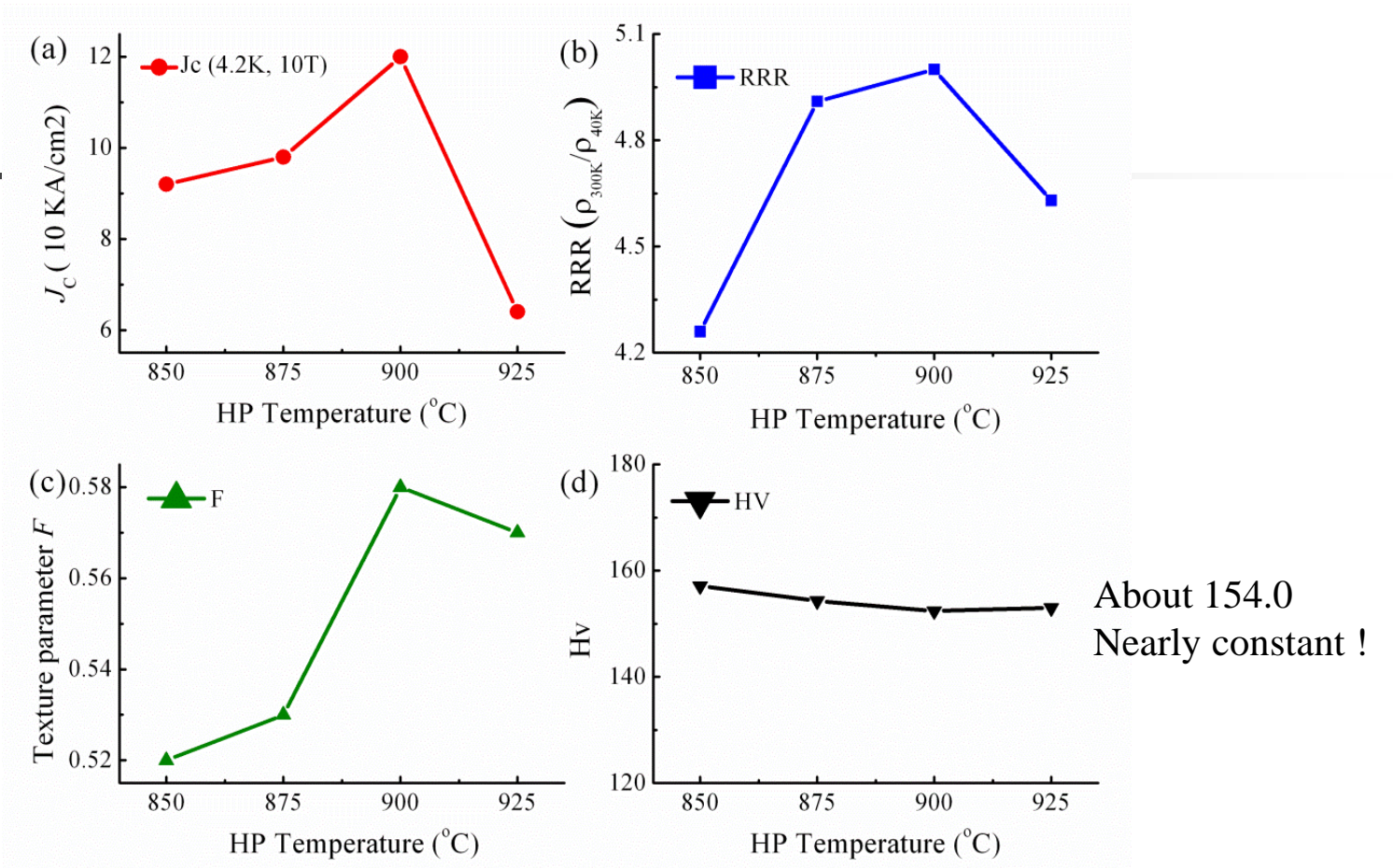
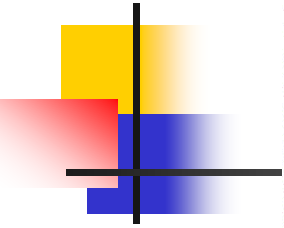
**Later achieved**

At 10 T,  $J_c = 1.2 \times 10^5$  A/cm<sup>2</sup>  
even in 14 T,  $J_c = \sim 10^5$  A/cm<sup>2</sup>

The superior  $J_c$  can be attributed  
to higher grain texture and  
improved densification.

Zhang et al., *APL* 104 (2014) 202601  
Lin et al., *Sci. Rep.* 4 (2014) 6944

# Reasons for high transport $J_c$ in HP900 tapes

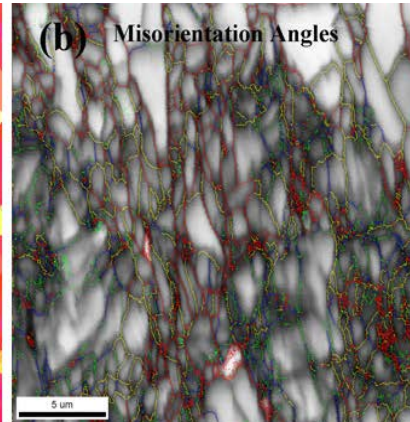
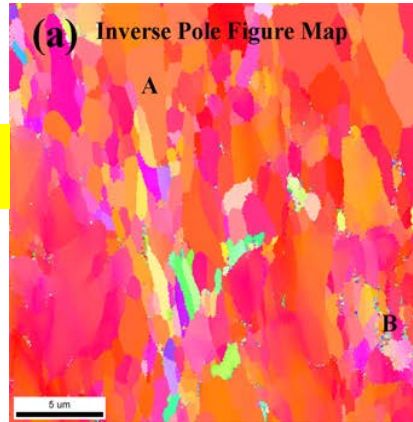


- The variation tendency of  $J_c$  values was qualitatively similar to those of  $F$  and RRR values.
- The hardness was almost saturated as soon as the hot pressing was applied.
- The  $J_c$  increase for HP900 tapes was mainly attributed to higher degree of c-axis texture and enhanced grain connectivity (high density).

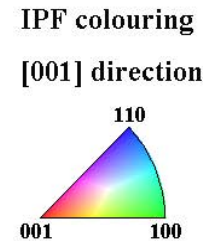
# EBSD images: the orientation mapping of grains

A useful tool to clarify the grain size, local orientation of the grains and misorientation angles between grains.

Upper: HP850



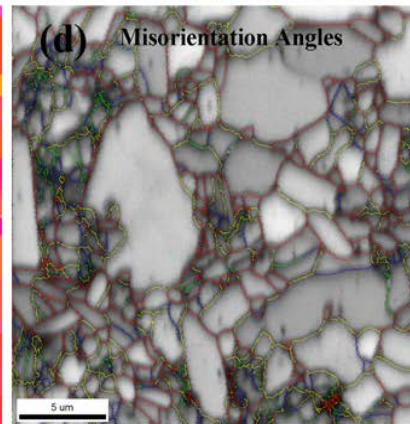
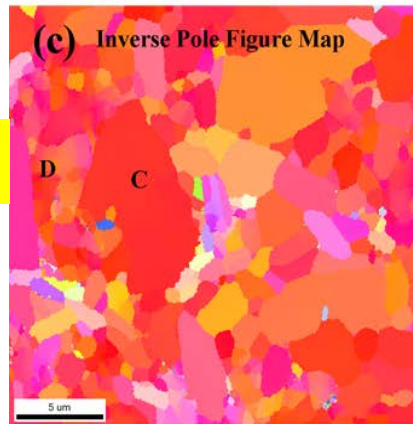
a/b plane



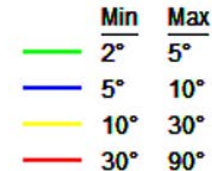
Inhomogeneous distribution of grain size



Lower: HP900



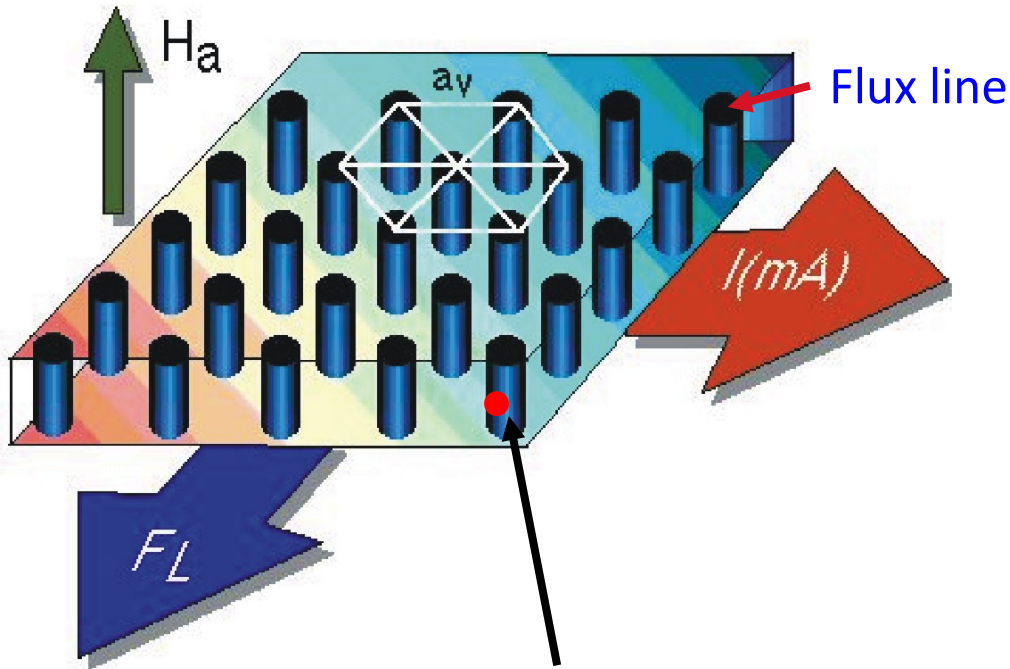
Misorientation Angles



A room for the  $J_c$  improvement

- The dominant orientation is (001) as the expected red color for both tapes, but there is a small (100) orientation for HP850 tapes as the green color.
- The large fraction of small misorientation angles between 2–10°C (HP850 tapes 23.3%, HP900 tapes 26.2%).

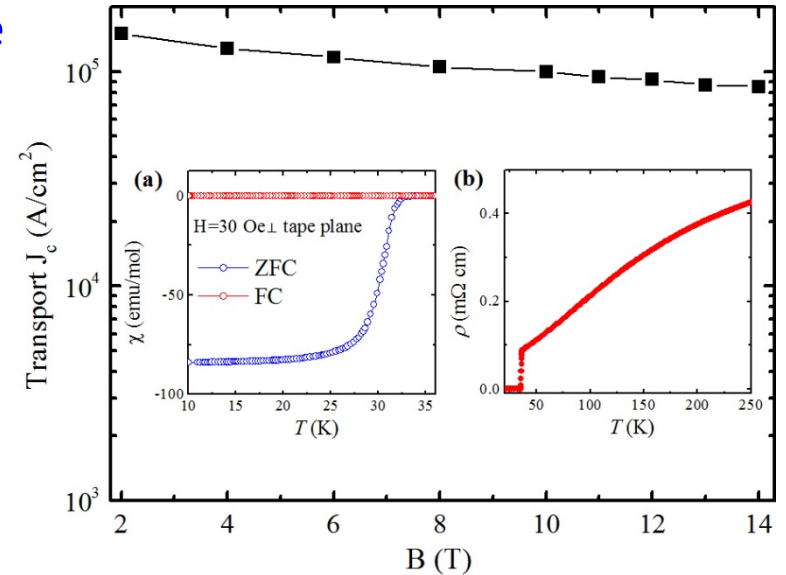
# Flux pinning and dynamics in hot-pressed high $J_c$ Sr122 tapes



**Pinning sites:** crystal defects, dislocations, nano-particles *et al.*

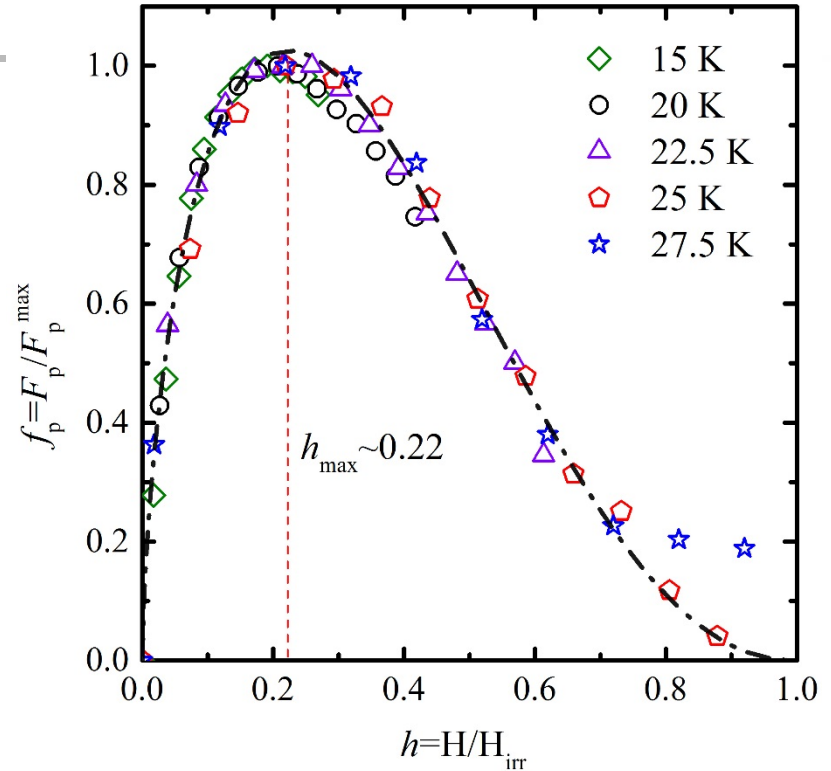
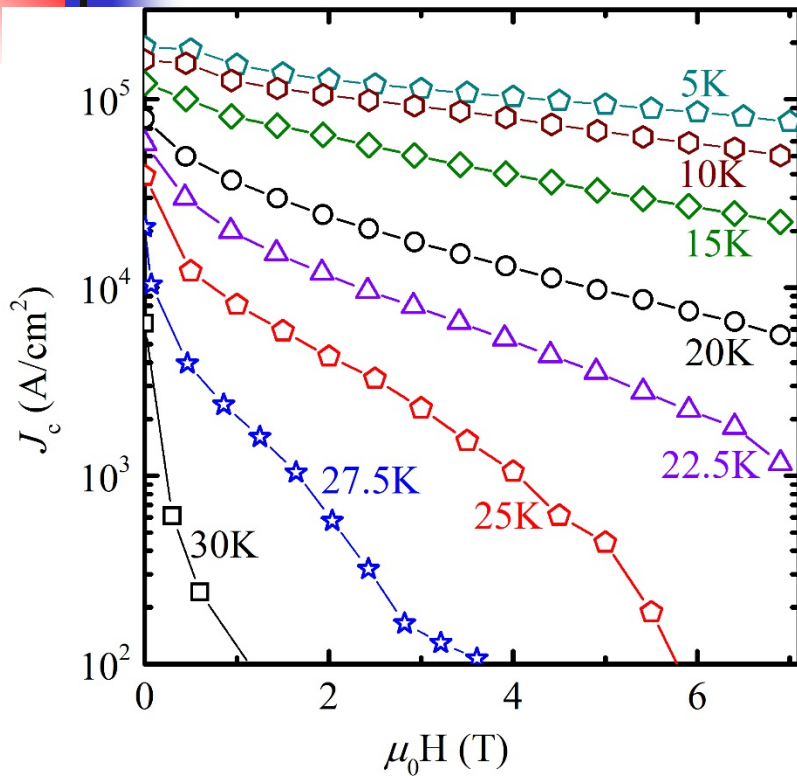
- ❑ No pinning, no  $J_c$ .
- ❑ Thermal excitation, quantum tunneling and mechanical vibration will lead to redistribution of flux and hence relax of magnetization.

*Flux pinning and motion are important factors that control  $J_c$*



- ❑  $J_c$  (4.2 K, 10 T)  $\sim 10^5$  A/cm<sup>2</sup>,  $I_c$  (4.2 K, 10 T)  $\sim 300$  A.
- ❑  $T_c \sim 35$  K, sharp superconducting transition.
- ❑ Residual resistivity ratio:  $RRR = \rho(250 \text{ K}) / \rho(37 \text{ K}) = 4.81$ , indicating good connectivity between grains.

# Flux pinning mechanism

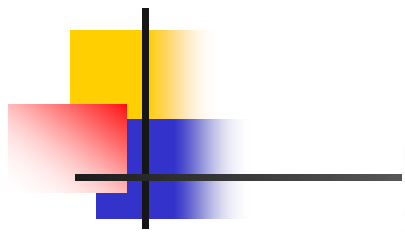


- ❑ Magnetic  $J_c$  is very close to the transport  $J_c$ , confirming that granularity is negligible.
- ❑ Dew-Hughes model:  $f_p = Ah^p(1-h)^q$ ,  $H_{max} = p/(p+q) = \mathbf{0.22}$

*Grain boundaries* and *dislocations* are dominant, *point pinning* also plays a role.

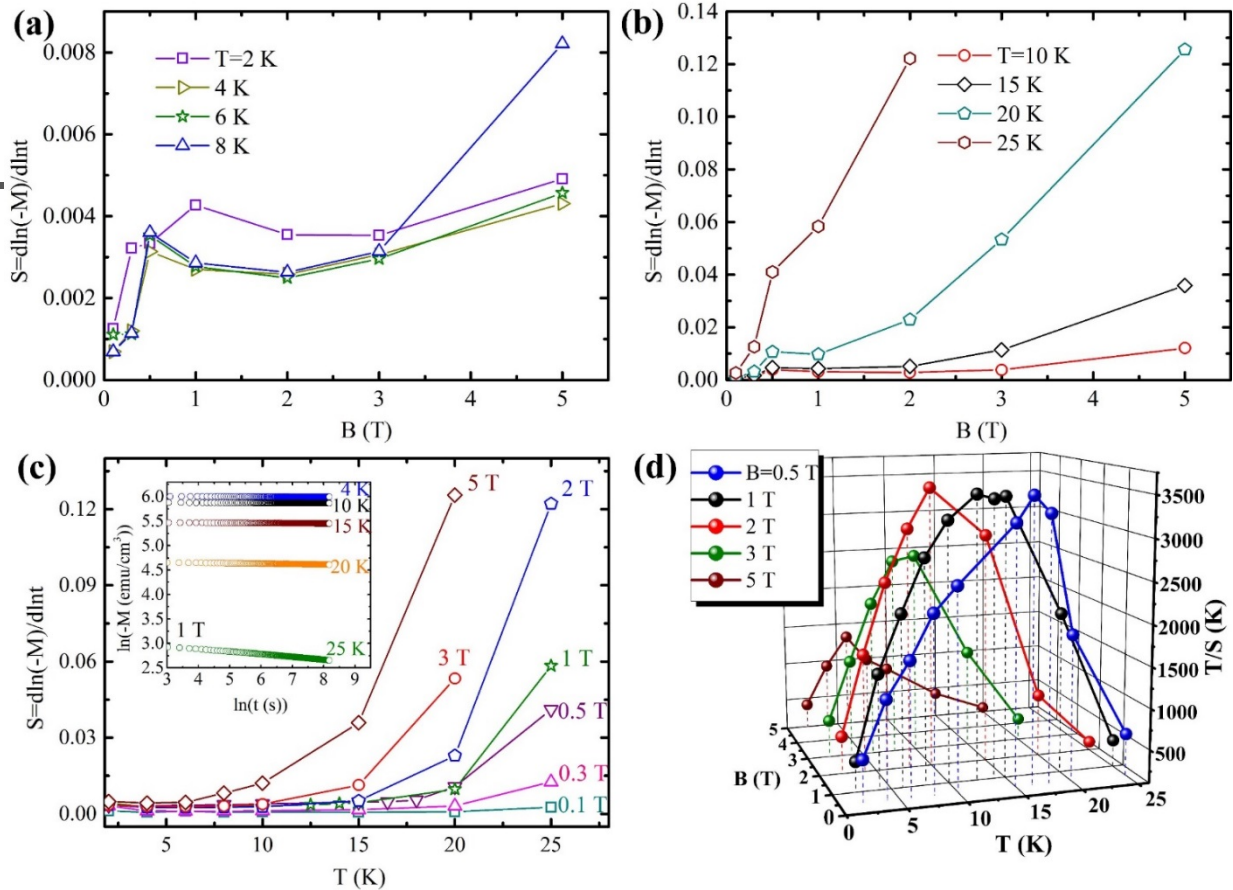
- ❑  **$\delta l$  pinning**, flux is pinned via spatial fluctuation of charge carrier mean free path.

# Magnetization relaxation - flux creep study



Relaxation rate:

$$S = d \ln(-M) / d \ln(t)$$

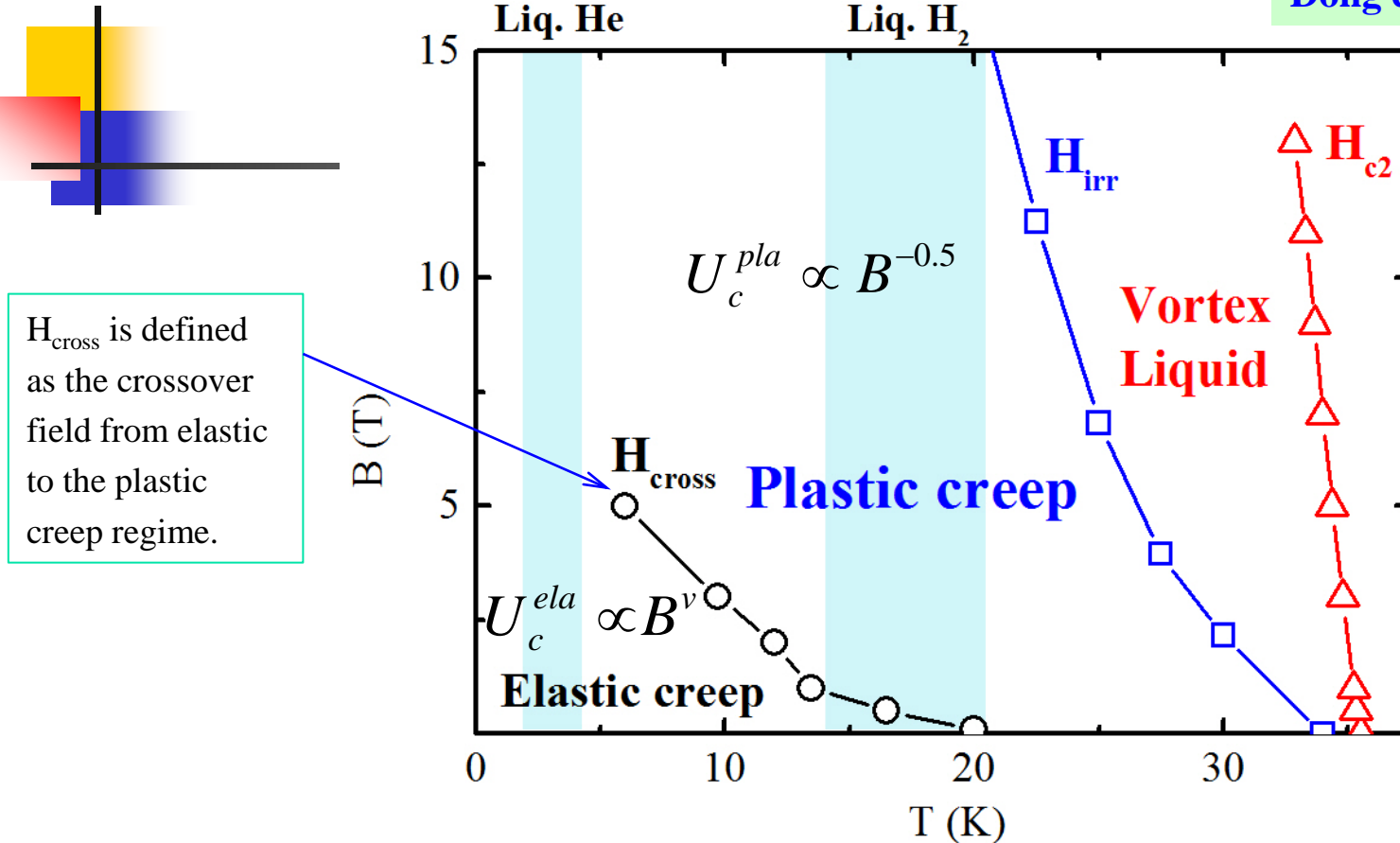


- ❑ Magnetic relaxation data indicates a logarithmic dependence of magnetization on time, *thermal activated flux creep*.
- ❑ Very **small relaxation rate** with **weak temperature and field dependence**, indicating strong flux pinning and weak field dependence of  $J_c$ .
- ❑ **Crossover** from elastic creep to plastic creep.



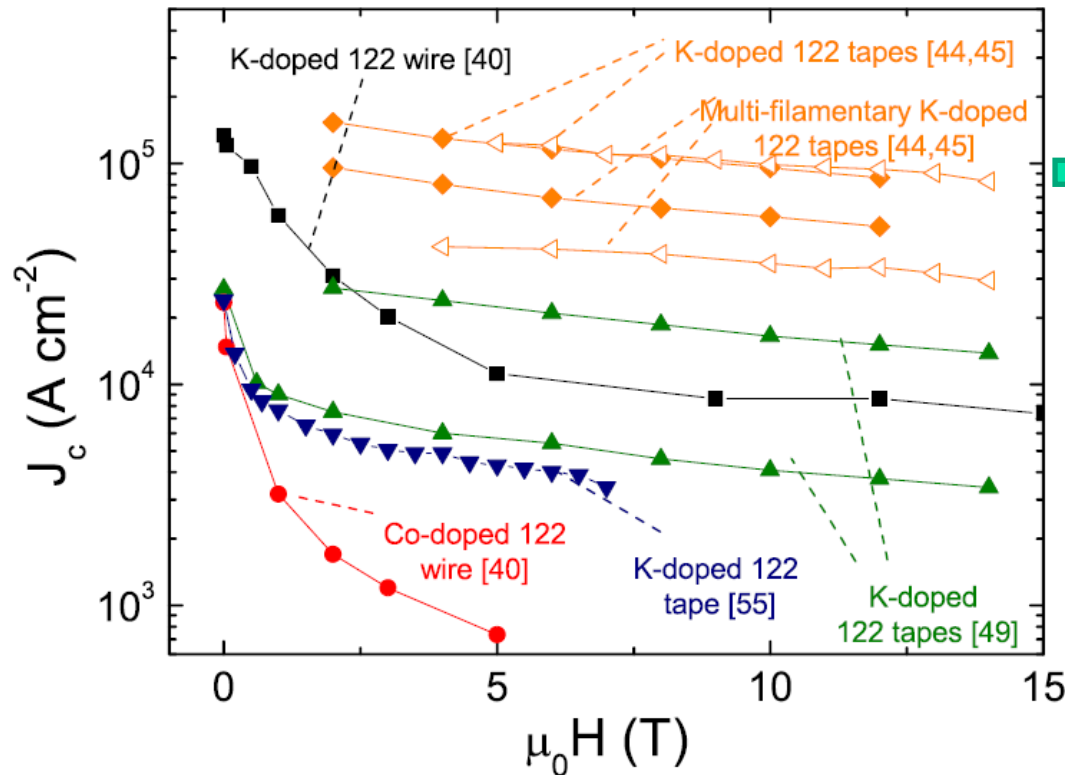
# Vortex phase diagram of high- $J_c$ HP-122 tapes

Dong et al., *JAP*, 2016



- More robust field dependence of  $J_c$  in the elastic creep regime.
- Weak field dependence of  $J_c$  in the liquid helium region, but  $J_c$  quickly decrease in the liquid hydrogen region.
- **To further increase flux pinning force:** i) decrease grain size to make more grain boundaries, ii) increase point pinning sites, *e.g.* radiation or nano-particle inclusion.

# Progress of $J_c$ values in 122 wires and tapes



4.2K, 10 T

$J_c = 1.2 \times 10^5 \text{ A/cm}^2$

IEECAS

Highest  $J_c$   
value reported  
in literature!

Putti, *SuST* 28 (2015) 114005

Next target:  $10^5 \text{ A/cm}^2$  at 20-30T

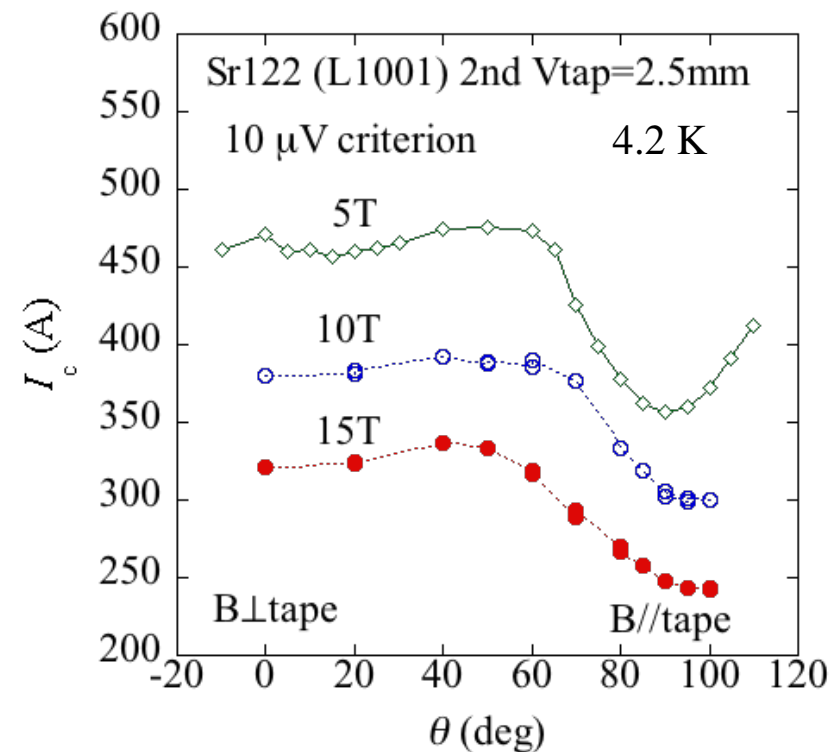
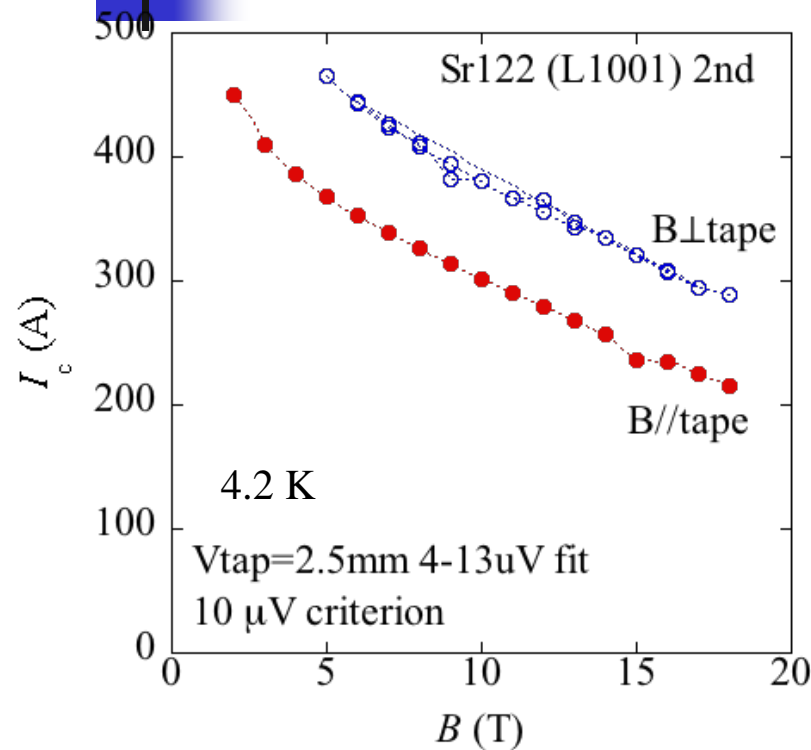
# Conductor requirements for practical applications

- Overall current density  $J_{cE}$  of conductor, not just of superconductor
- Performance in field
- Filamentary architecture essential for AC applications
- Anisotropy of  $J_{cE}$  with respect to field direction

- Cost
  - Conductor itself
  - Cooling
- Scaleability of fabrication
- Mechanical
  - Strength, bend radius, .....
- Conductor shape
  - tape or wire

# $J_c$ properties at 4.2 K for HP Sr-122/Ag tapes

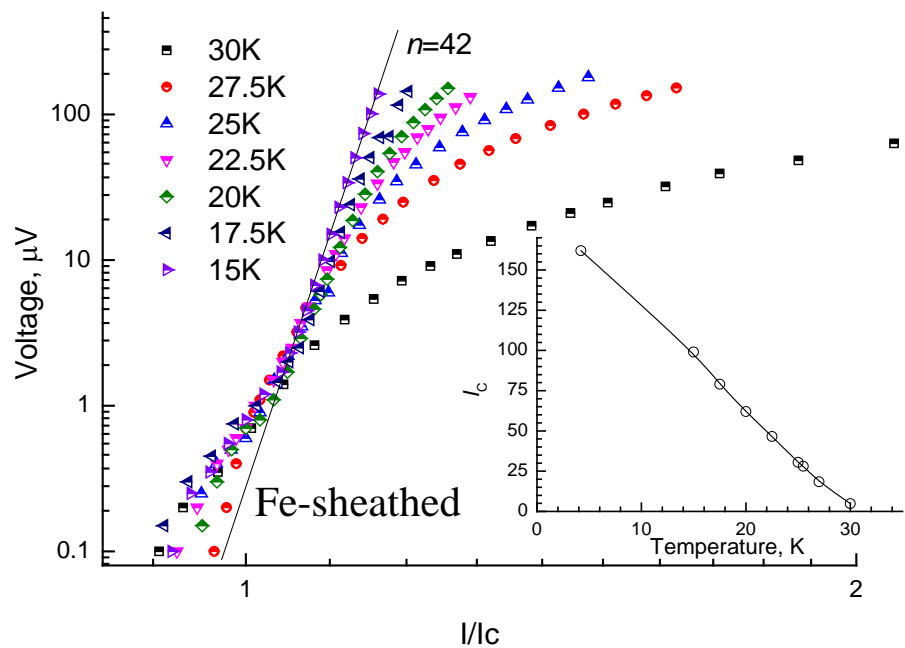
-- Measured by Dr. S. Awaji  
HFLSM, Tohoku Univ.



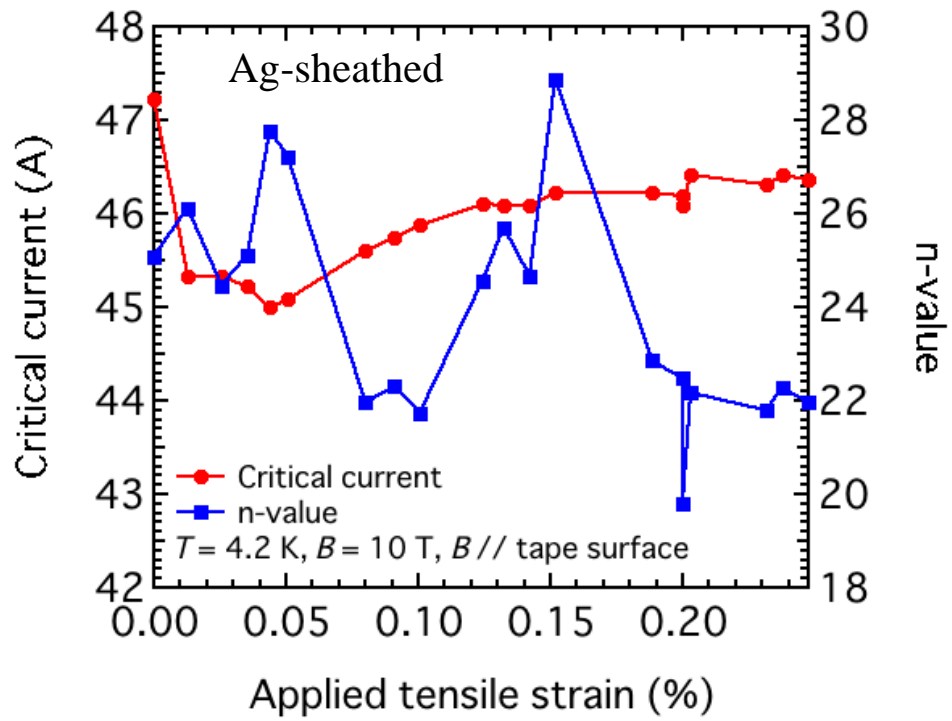
- ◆ The  $I_c$  in applied magnetic fields is slightly higher in the perpendicular field ( $I_c^\perp$ ) than in the parallel field ( $I_c^\parallel$ ).
- ◆ The anisotropy ratio ( $\Gamma = I_c^\perp / I_c^\parallel$ ) is less than **1.5**, quite small, very promising for applications.

# Temperature dependence of $n$ value for Sr-122 tapes

-- Measured by Prof. Yang  
Univ. of Southampton, UK



-- Measured by Dr. Oguro  
HFLSM, Tohoku Univ., JP

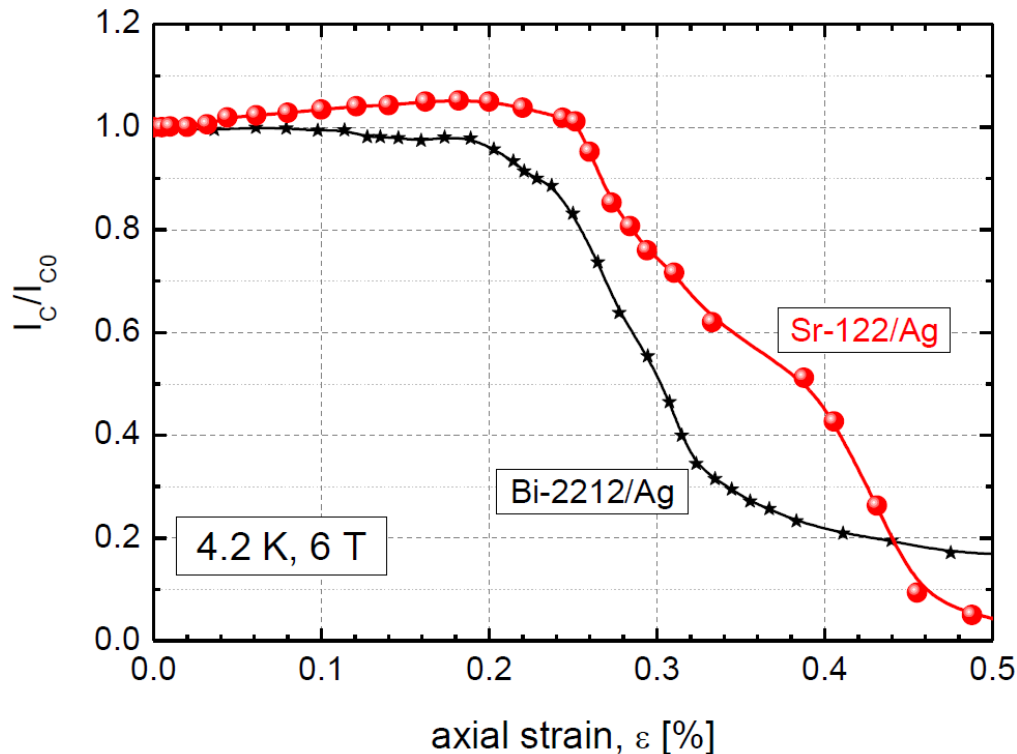


**At 20 K, the  $n$  value was over 30**

**At 4.2 K, the  $n$  value was over 20**

# The first strain measurements of Sr-122/Ag tapes

-- Measured by Dr. Kovac  
Slovak Academy of Sciences



At 4.2 K, 10 T:  $I_c > 125A$

Irreversible strains:

$\varepsilon = 0.25\%$



which seems better than that  
of Bi-2212/Ag

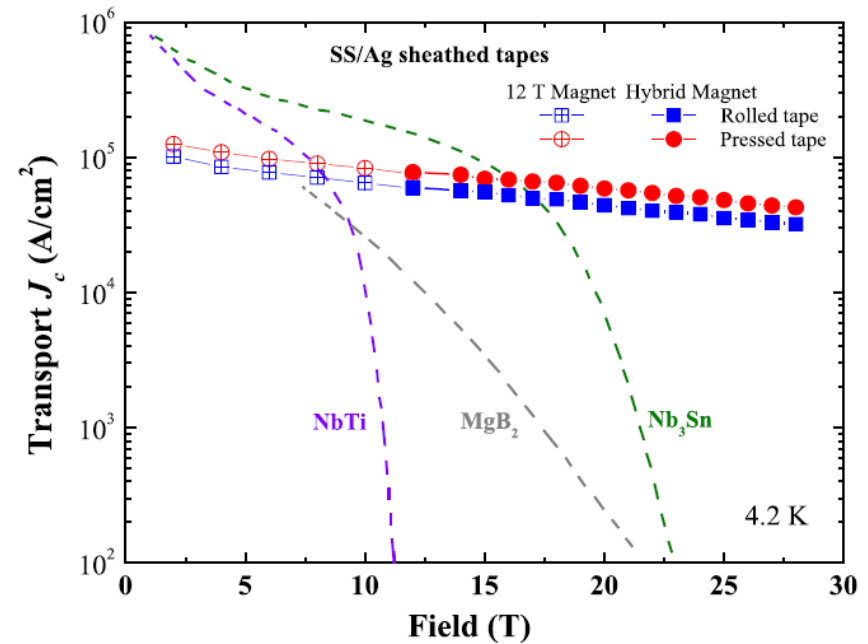
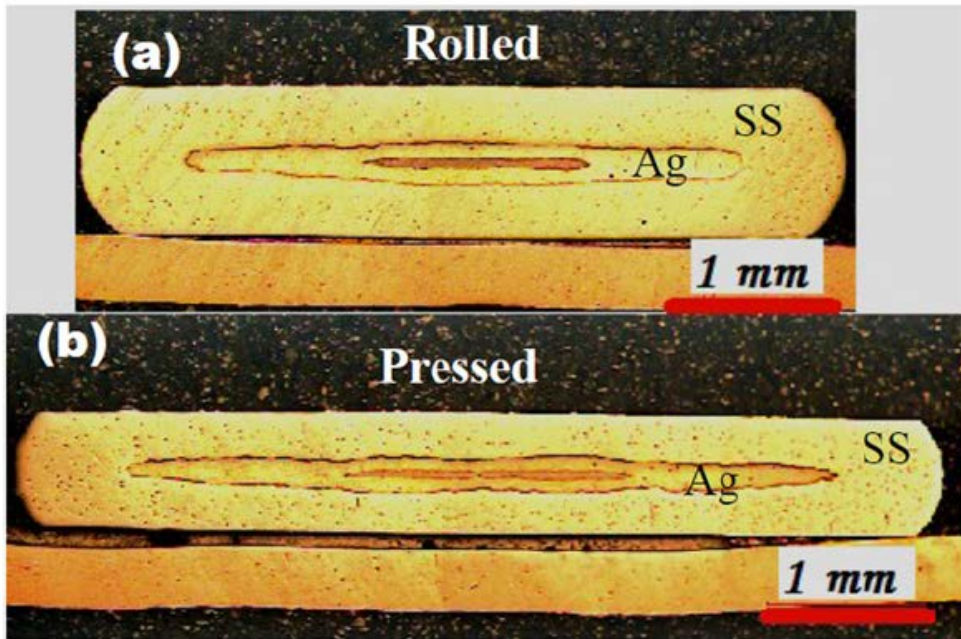
The first observation of strain effects  
on the critical current of 122 wires

Next step:

Improvement of mechanical property of pnictide wires will be one of  
the major challenges for high field applications

# Fabrication of stainless steel/Ag double sheathed Ba122 tapes

Highly mechanical property is expected!

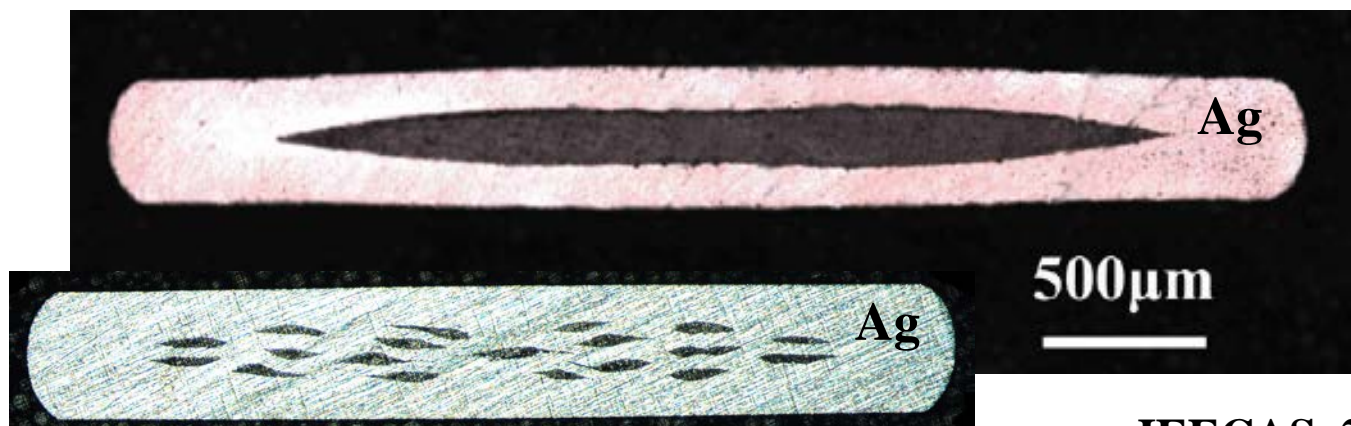


High  $J_c$ , but show lower  $J_e$

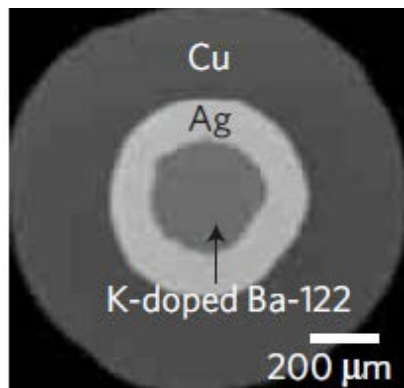
# So far, all high- $J_c$ pnictide wires and tapes were made by using Ag as sheath material

Ag is very expensive

We should find other cheap materials, in order to reduce the cost!



IEECAS, 2014



Florida, 2012



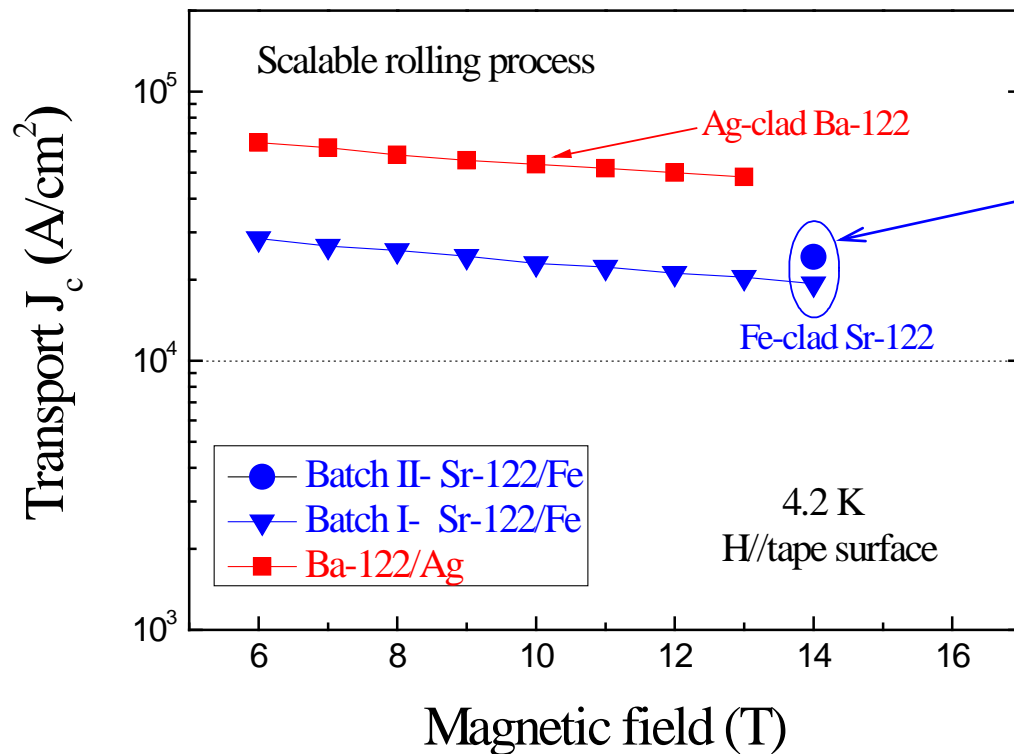
NIMS, 2014



# Fabrication of Fe-cladded 122 tapes

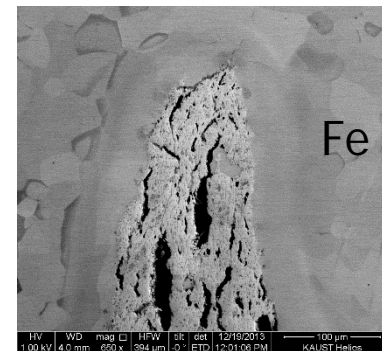
-- tape thickness=0.6 mm

-- by the scalable rolling



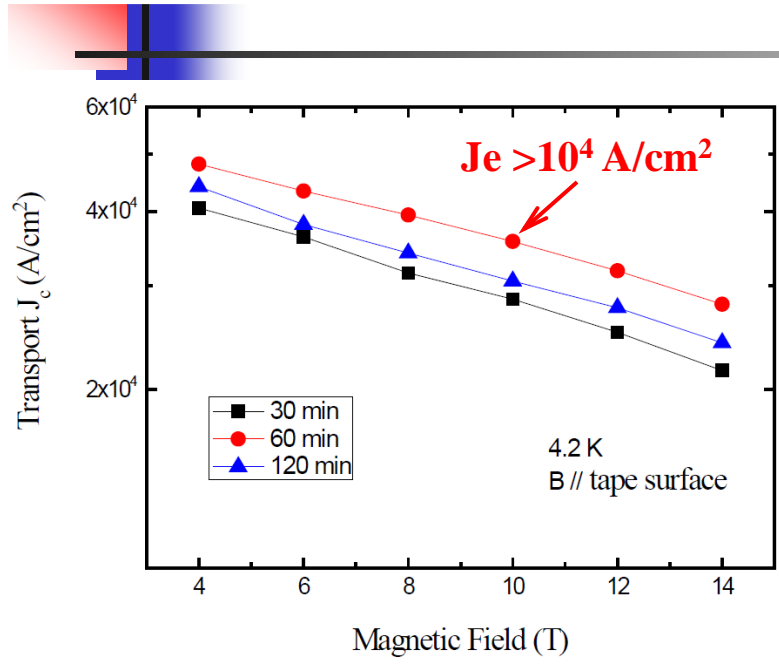
At 4.2 K and 14 T:

$$J_c = 2.4 \times 10^4 \text{ A/cm}^2$$

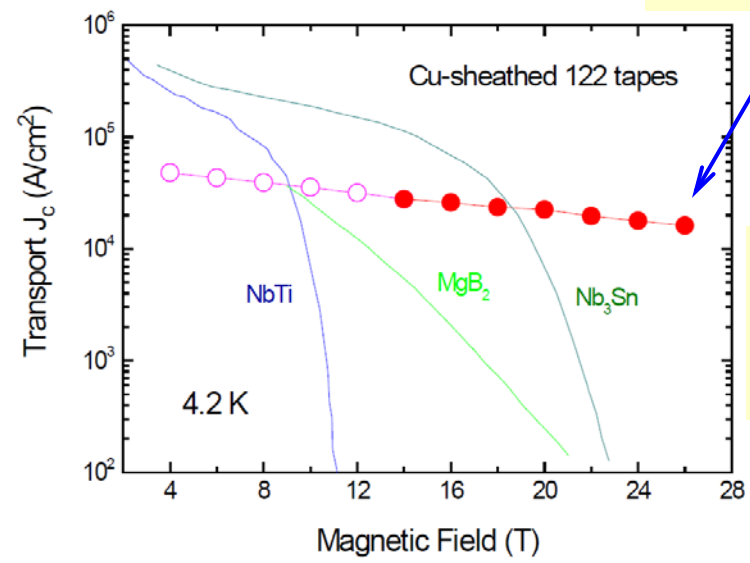


From an economic point of view, the Fe sheath is more attractive than the Ag sheath in fabricating Sr122/Ba122 tapes for practical applications.

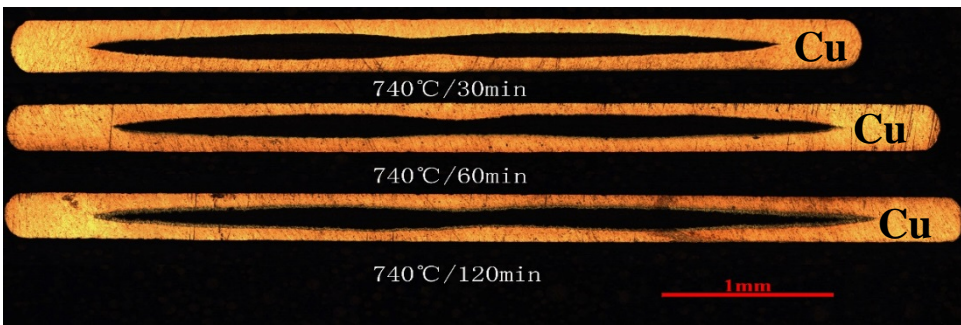
# High $J_c$ in Cu-sheathed Sr-122 tapes at low temperature 740°C



At 26 T:  
 $J_c = 1.6 \times 10^4$  A/cm $^2$

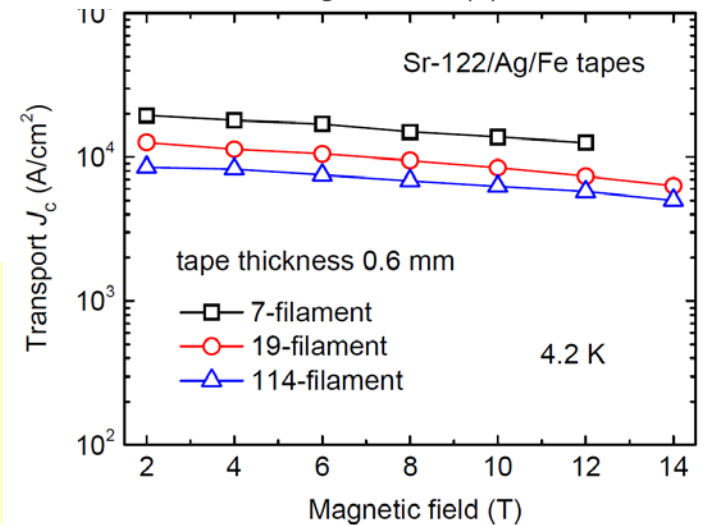
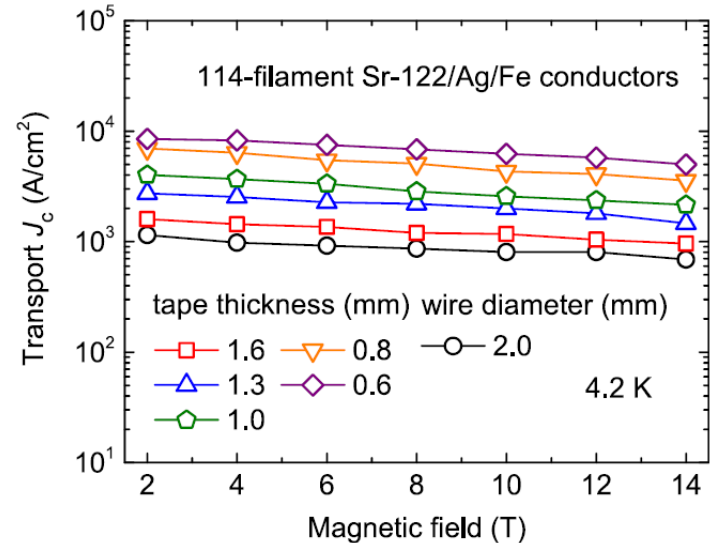
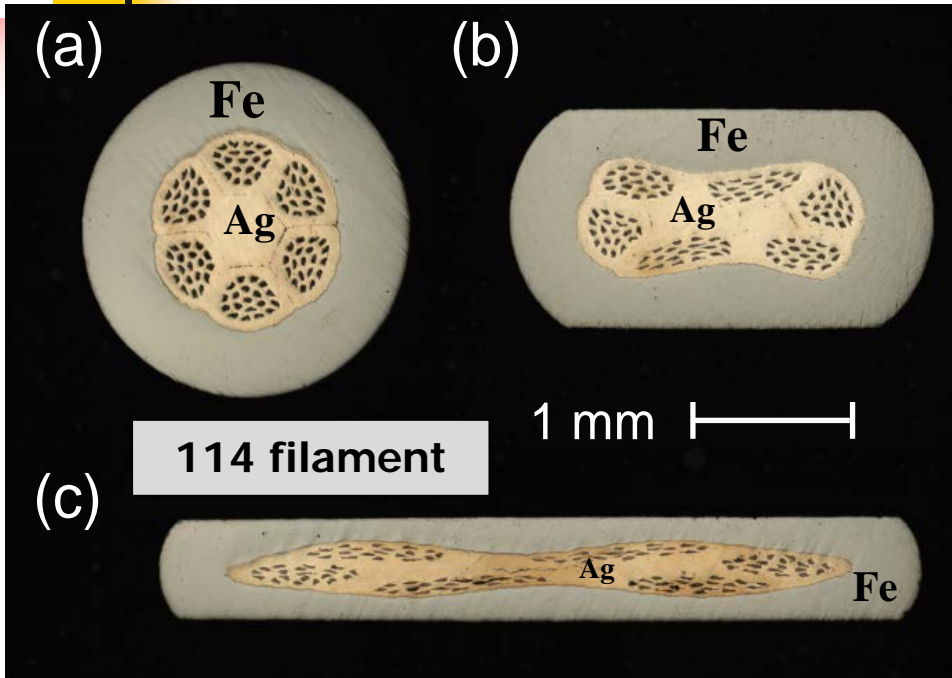


At 4.2 K, 10 T  
**Engineering  $J_e$ :**  
 $> 10^4$  A/cm $^2$



- ◆ The HP740/60 tape has a uniform and well connective superconducting phase and only a small amount of diffusion of Cu.
- ◆ The best transport  $J_c$  reaches  $3.5 \times 10^4$  A/cm $^2$  at 10 T and keeps  $1.6 \times 10^4$  A/cm $^2$  at 26 T.

# Fabrication of 114-filament Sr-122/Ag/Fe wires by the drawing and rolling



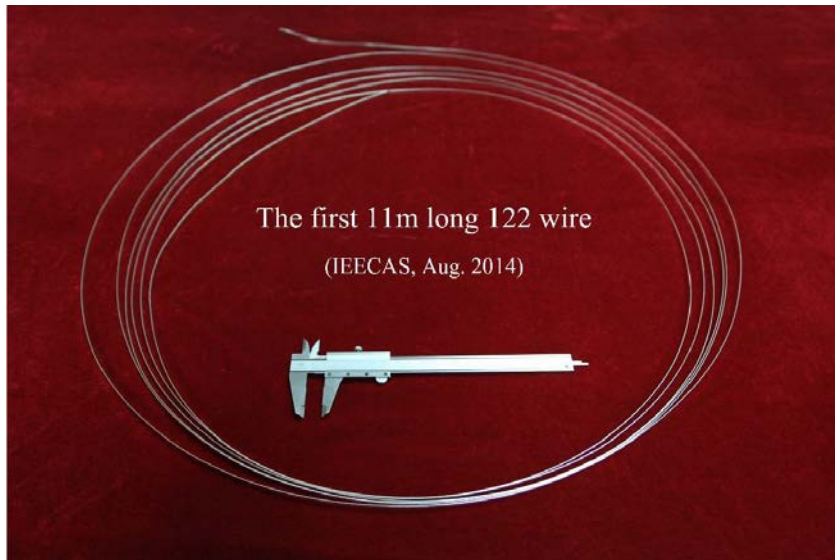
**At 4.2 K, 10 T:**

- ◆ 114-core round wires:  $J_c=800$  A/cm<sup>2</sup>.
- ◆ When they are flat rolled into tapes, the  $J_c$  grows with the reduction of tape thickness. the  $J_c=6.3 \times 10^3$  A cm<sup>-2</sup> in 0.6 mm thick tapes.
- ◆ 7-core tapes:  $J_c=1.5 \times 10^4$  A/cm<sup>2</sup>.
- ◆ This  $J_c$  degradation can be ascribed to the sausage effect.

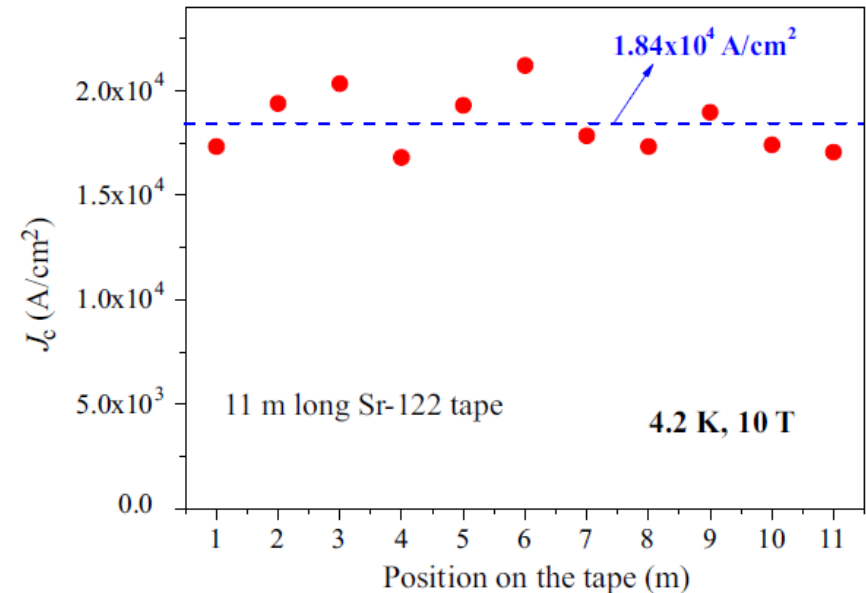
# The first 11m long Sr-122 tape

-- by the scalable rolling process

The first long wire-- 11 m



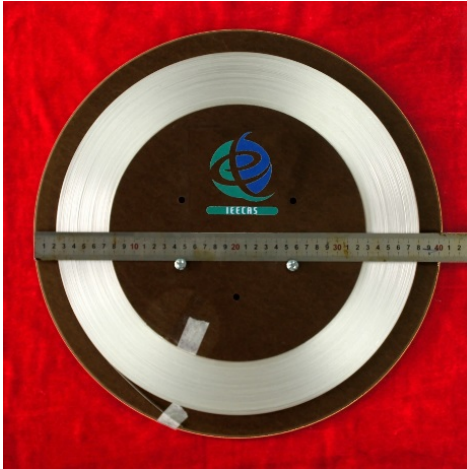
Uniform wires can be fabricated



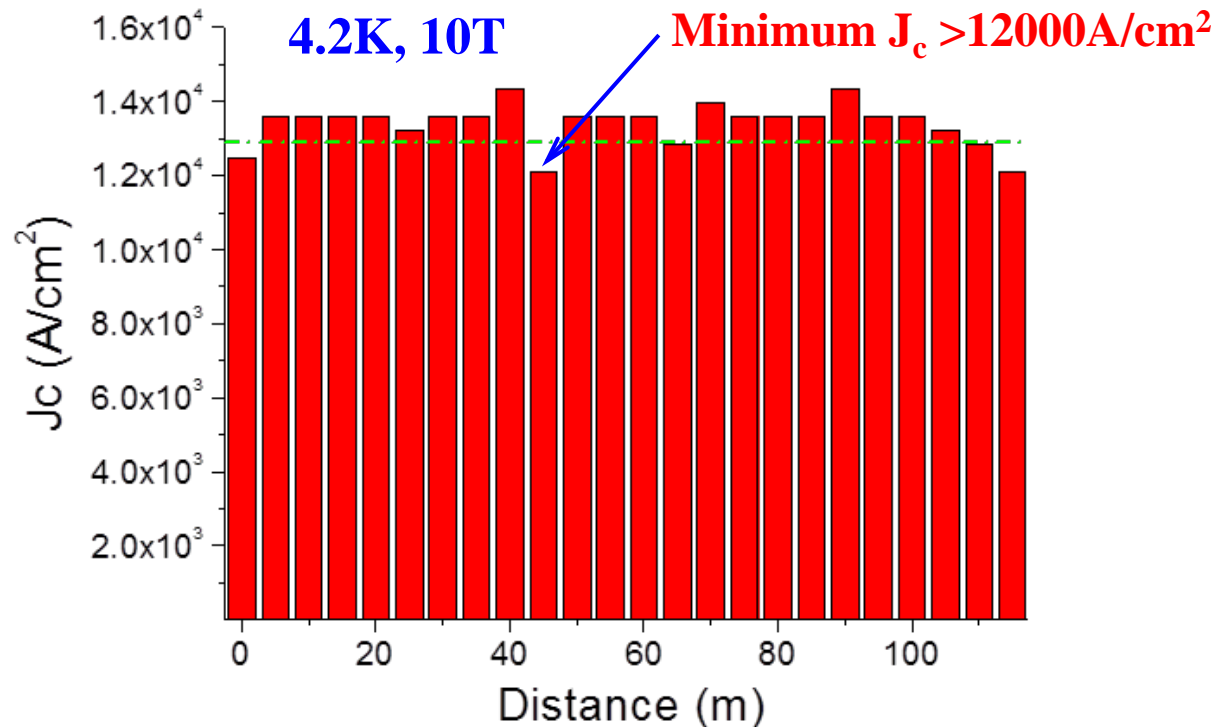
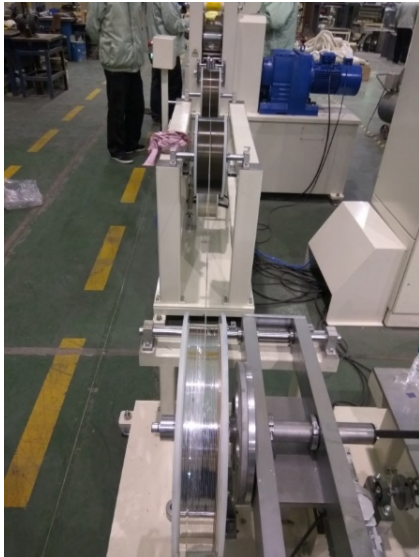
The average  $J_c$  of this long Sr122/Ag wire is  $\sim 18400 A/cm^2$   
The fluctuations of the  $J_c$  is  $\sim 5\%$

# The world's first 100 meter-class iron-based superconducting wire

-- Presented at ASC2016, Denver



115 m long 7-filamentary wire



At 4.2 K, 10 T, transport  $J_c$  distribution along the length of the first 100 m long 7-filament Sr122 tape

<http://snf.ieceesc.org/pages/new-paper-and-result-highlights>



# Challenges for the next stage R&D

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## ✓ **Ultra High In-Field Critical Currents: $I_c - B$**

→ e.g. engineering current density  $J_e > 500 \text{ A/mm}^2$  @ 4.2 K, 20 T

## ✓ ***Homogeneous long length wires:***

→ High performance, high productivity, length up to 1 km level

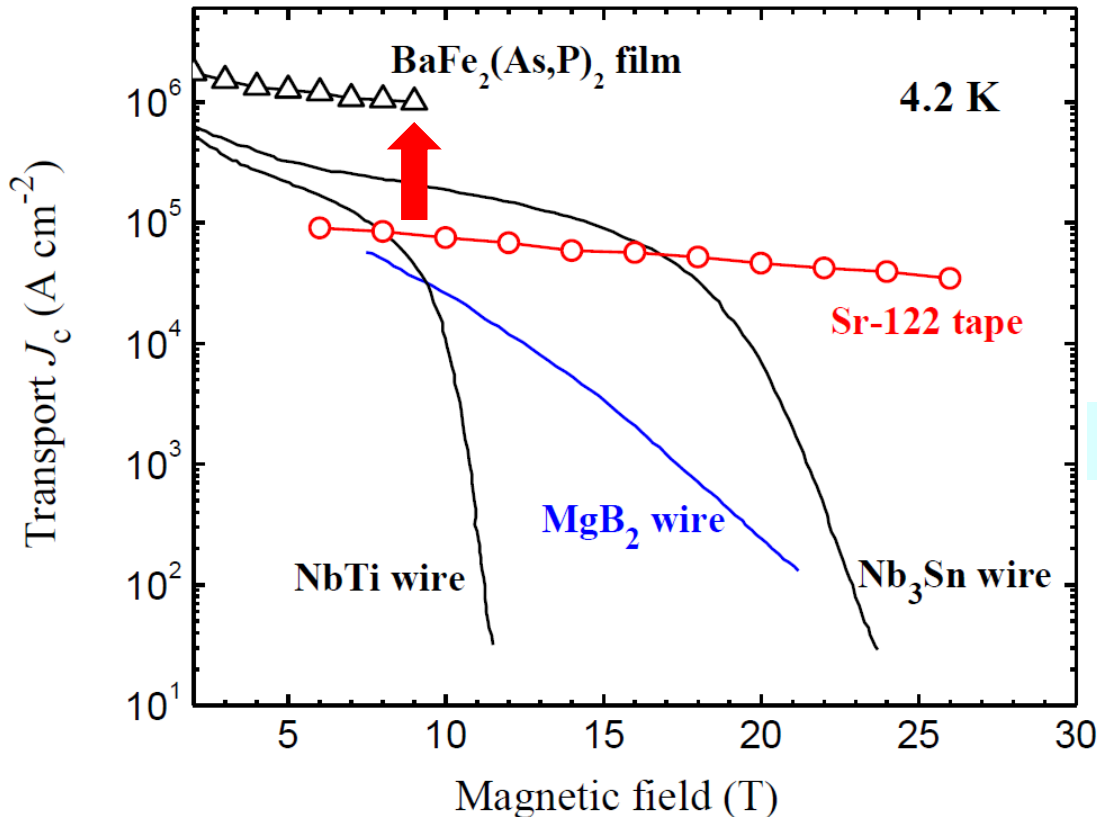
## ✓ ***Low Cost Wires:***

→ e.g. Cu- or Fe-sheathed wire fabrication, instead of using Ag

## ✓ ***High Mechanical Strength Wires:***

→ *Tensile, Bending*

# Prospects



At present

$J_c > 10^5$  A/cm<sup>2</sup> at 10 T, 4.2 K

The  $J_c$  is much lower than the pair breaking current density  $> 10^8$  A/cm<sup>2</sup>.

Much room to further improve  $J_c$ !

Strong promising for high-field applications !

- ◆ We believe that iron-based wires would be possible to operate at 4.2 K  $> 20$  T or 20-30 K at  $> 10$  T.
- ◆ An scalable process is required to fabricate high performance long length tapes, e.g., *Rolling (hard sheath), Hot Rolling or Hot isostatic press (HIP)*...

## **Contributors:**

**Xianping Zhang, Chao Yao, He Lin, Chiheng Dong, Qianjun Zhang,  
Dongliang Wang, He Huang**

**Institute of Electrical Engineering, CAS**

## **Collaborators:**

**S. Awaji, K. Watanabe**

**Institute for Materials Research, Tohoku University, Japan**

**Hai-hu Wen**

**Nanjing University, China**

**Jianqi Li**

**Institute of Physics, Chinese Academy of Sciences**

**Xiaolin Wang, S. X. Dou**

**Wollongong University, Australia**

**P. Kovac**

**Slovak Academy of Sciences, Slovakia**

**Yifeng Yang**

**University of Southampton, UK**





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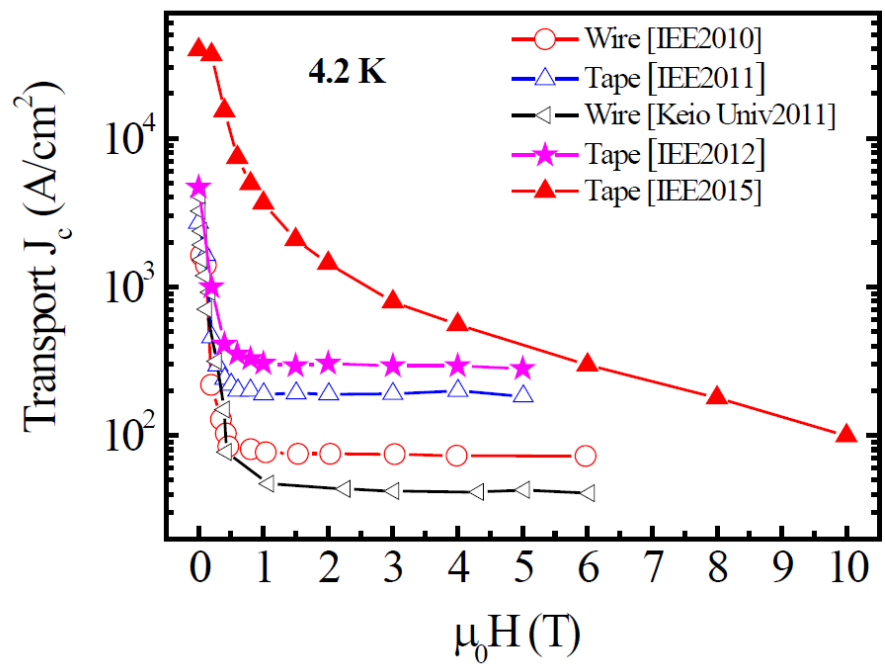
*Thank you for your attention*



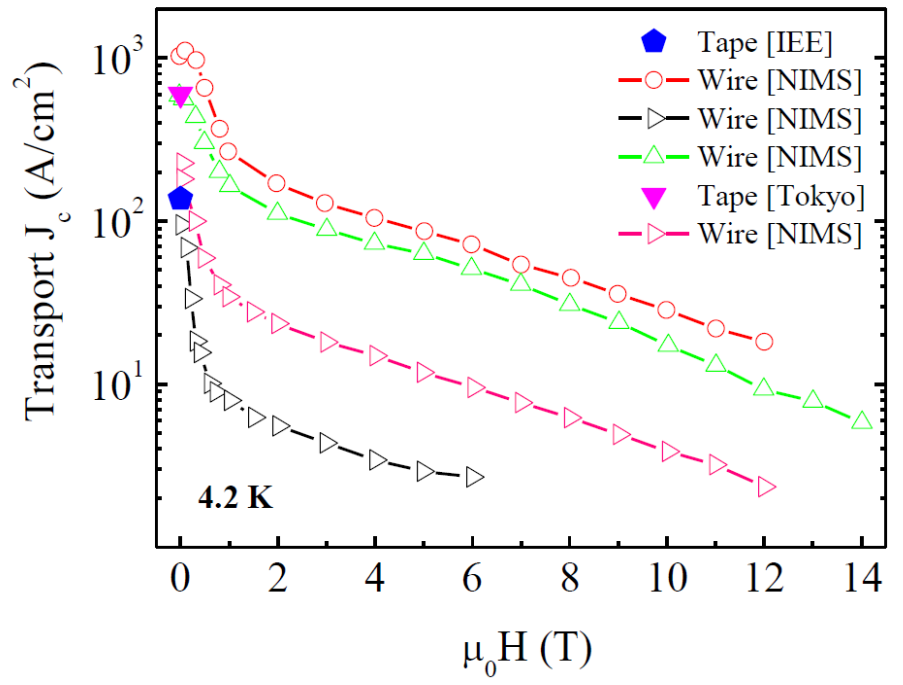
122 tapes showed the highest  $J_c$ :  $10^5$  A/cm<sup>2</sup> @ 10 T, 4.2 K

# 1111 and 11 wire and tapes: $J_c \sim 200$ A/cm<sup>2</sup> in high fields

1111 type -  $\text{Sm}[\text{O}_{1-x}\text{F}_x]\text{FeAs}$



11 type -  $\text{FeTe}_{1-x}\text{Se}_x$



- ◆ The  $J_c$  values obtained are still two to three orders of magnitude lower than for the 122 tapes.
- ◆ **1111 wires:** how to control fluorine content during sintering.
- ◆ **11 wires:** hard to remove excess Fe.