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$NiCo_2O_4$ nanowires@MnO_x nanoflakes supported on stainless steel mesh with superior electrocatalytic performance for anion exchange membrane water splitting



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ABSTRACT

A core-shell structured catalyst with NiCo₂O₄ nanowires as the core and MnO_x nanoflakes as the shell is synthesized and deposited on a stainless steel mesh, and applied in anion exchange membrane water splitting. This unique structured catalyst allows the core material, NiCo₂O₄ nanowires, to serve as the electrically conductive skeleton, and the shell material, MnO_x nanoflakes, to be the electroactive sites. As a result, the catalyst exhibits a superior electrocatalytic activity ($\eta = 342 \text{ mV} @10 \text{ mA cm}^{-2}$) toward the oxygen evolution reaction. The water splitting cell with the catalyst can be operated at 200 mA cm⁻² with an overpotential of about 500 mV and stably run for 100 h without significant degradation. The results demonstrated here shows that the NiCo₂O₄@ MnO_x is a promising candidate for the oxygen evolution reaction in the alkaline media.

1. Introduction

Recently, anion exchange membrane (AEM) water splitting has been developed to address the cost issue related to its counterpart, proton exchange membrane water splitting [1–3]. Although the alkaline environment permits the employment of non-precious catalysts as the catalyst for oxygen evolution reaction (OER), the rational design of an efficient electrode with high electroactivity and excellent stability is still a critical challenge for AEM water splitting [4–6]. As one of the most promising alternatives, novel catalysts based on the transition metal oxides (TMOs) with different compositions and morphologies are widely explored. Among these TMOs, NiCo₂O₄ has been extensively studied as the catalyst for OER, owing to its superior electrical conductivity, abundant and tunable active sites [7–9].

Meanwhile, the NiCo₂O₄ serves as the skeleton to deposit other nanostructured catalysts, such as Co₃O₄/NiCo₂O₄ [10], graphene nanosheets/NiCo₂O₄ [11] and Au/NiCo₂O₄ [12], has been recently proposed to further improve the electrocatalytic activity. Even though there are some promising results based on the NiCo₂O₄ have been achieved, the overpotentials at a current density of 10 mA cm⁻², which is a metric relevant to solar fuel synthesis, in these composite catalysts are still high, which restricts the application of these catalysts in the AEM water splitting.

To address this issue, we electrochemically deposited MnOx

nanoflakes (NFs) on the NiCo₂O₄ nanowires (NWs)-decorated stainless steel (SSL) mesh and explored its electrocatalytic activity toward the OER in the alkaline media for the first time. The MnOx was chosen due to the nanostructured manganese oxides have been shown as a superior catalyst for the OER in the alkaline media [13]. The superior electrocatalytic activity was not only demonstrated in the three-electrode-cell system but also verified in an AEM water splitting cell.

2. Experimental

2.1. Synthesis of NiCo₂O₄@MnO_x supported on SSL mesh

A twill-woven stainless steel mesh (SSL mesh, #600) was selected as the supporting layer. The self-supported NiCo₂O₄ NWs on the SSL mesh were prepared as reported elsewhere [14]. The mass loading was 0.5 mg cm^{-2} . After that, MnO_x NFs were fabricated on NiCo₂O₄ NWs by an electrodeposition process in a conventional three-electrode cell [15]. The deposition was performed in a solution containing 0.1 M manganese acetate and 0.1 M ammonium acetate. The NiCo₂O₄/SSL was vertically submerged into the solution as the working electrode. Pt mesh and Ag/AgCl (3.5 M KCl) were used as the counter electrode and reference electrode, respectively. The MnO_x precursors were deposited galvanostatically with a current density of 2 mA cm⁻² and then annealed at 300 °C in ambient air for 1 h to obtain NiCo₂O₄@MnO_x/SSL.

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The mass loading of MnO_x was 0.6 mg cm⁻².

2.2. Material characterizations

Transmission electron microscopy (TEM) images were obtained by operating a high-resolution JEOL 2010F TEM system with a LaB₆ filament at 200 kV. Surface morphology was determined by a field-emission scanning electron microscope (FESEM, JEOL 6700F). The X-ray diffraction (XRD) pattern was analyzed with a Philips high-resolution X-ray diffraction system (model PW 1825) using a Cu K α source operating at 40 kV with the scan rate of $0.025^{\circ} \, {\rm s}^{-1}$. The X-ray photoelectron spectroscopy (XPS) characterization was determined by a Physical Electronics PHI 5600 multi-technique system using Al monochromatic X-ray at a power of 350 W. Raman measurements were performed with an RM3000 (Renishaw) micro-Raman spectrometer at 514.5 nm. The laser beam was focused onto the sample with a 50 × objective.

2.3. Electrochemical measurements

The electrochemical activity was investigated by linear sweep voltammetry (LSV) that implemented with the same three-electrode cell as described above. The samples were subjected to electrochemical treatment by potential scanning from 0 to + 0.70 V vs. Ag/AgCl in 0.1 M KOH. The electrolyte was saturated with oxygen by bubbling oxygen gas at least 30 min in advance. A home-made IrO₂/C (10 wt%) was employed as the benchmark for OER process. The IrO₂/C with a mass loading of 0.2 mg cm⁻² was brushed on the SSL mesh with a home-made ionomer as the binder.

2.4. AEM water splitting

The AEM water splitting was evaluated in a home-made water electrolysis setup. The membrane with a thickness of 60 μ m was cast with the home-made ionomer suspension. The as-fabricated samples were directly employed as the electrode for OER (5 pieces) while a commercial powder catalyst (Acta 4030) was employed as the electrode for hydrogen evolution reaction (HER). The HER catalyst loading in the as-prepared electrode was 4 mg cm⁻². The performance of IrO₂/C supported SSL mesh was also evaluated for the purpose of comparison. The fabrication procedure of membrane-electrode-assemblies (MEAs) and test protocol were the same as reported in our previous work [16]. All measurements were performed at 60 °C.

3. Results and discussion

A typical low-resolution SEM image (inset of Fig. 1a) illustrates that the entire surface of the bare SSL mesh is densely covered with NiCo₂O₄ NWs. A high-resolution SEM image (Fig. 1a) reveals that NiCo₂O₄ NWs with a hexagonal cylinder architecture are radially grown on the SSL mesh with a length of 0.5–4 μ m and an average diameter of 50 nm. It is also clearly observed that the mesopores emerge in the NiCo2O4 NWs due to the stacking of small nanoparticles (5-20 nm). The interconnected mesopores will be beneficial for the accessibility to electrolyte and oxygen, enabling full utilization of the electroactive surface area of NWs and fast electronic, ionic transportation through the electrode. Subsequently, MnO_x NFs are electrodeposited galvanostatically on NiCo2O4 NWs followed by a thermal treatment process. As presented in Fig. 1b-c, the surface of NiCo₂O₄ NWs is completely deposited by various MnOx NFs with distinct folds (thickness: 3-9 nm) due to the highly electrical and electrochemical active property of NiCo₂O₄ NWs. Meanwhile, thanks to the weak alkaline electrolyte, the mesoporous structure of NiCo₂O₄ NWs is persevered after the electrodeposition process, as confirmed by a typical TEM image (Fig. 1d). Typical HRTEM image (Fig. 1e) shows that clear lattice fringes with interfringe distances of approximately 2.52 Å and 2.76 Å, corresponding to the (311) and (220) plane of the spinel phase NiCo₂O₄. Then, the selected area electron diffraction measurement was performed (not shown), and the result indicates that NiCo₂O₄ NWs possessed a polycrystalline structure. Meanwhile, typical HRTEM image presented in Fig. 1f shows that asfabricated MnO_x NFs possess a nonuniform microcrystalline structure.

The crystallinity of as-prepared samples is determined by XRD (Fig. 2a). The strong diffraction peaks, locating at 43.4°, 50.6° and 74.5°, are derived from the SSL mesh while other diffraction peaks emerged in the NiCo₂O₄/SSL sample, can be readily indexed as spinel NiCo₂O₄ phase (JCPDS card No. 073-1702). After fabricated with MnO_x NFs, no new peaks are visible in the NiCo₂O₄@MnO_x/SSL sample apparently, suggesting that MnO_x NFs exist as an amorphous or weak-crystallinity structure in the composite nanostructure, as confirmed by HRTEM. It should be noted that Mn₂O₃ is the best phase of MnO_x for the OER [15,17]. However, the transfer from MnOOH to Mn₂O₃ requires a thermal treatment at 500 °C in the ambient air, in which process the morphology of NiCo₂O₄ is completely collapsed. This is the reason why we thermally implemented the conversion at relatively low temperature. Actually, the MnO_x can be partially converted to Mn₂O₃ by the subsequently electrochemical measurement [13].

The surface elemental compositions are further investigated by XPS. The raw Co 2p spectrum of NiCo₂O₄/SSL can be deconvoluted into two peaks that are assigned to Co^{3+} (779.3 eV for Co $2p_{3/2}$) and Co^{2+} (780.8 eV), and one shakeup satellite (Fig. 2b). The raw Ni 2p spectrum of NiCo2O4/SSL can be also deconvoluted into two peaks that are assigned to Ni²⁺ (854.2 eV for Ni $2p_{3/2}$) and Ni³⁺ (855.8 eV), and one shakeup satellite (not shown). The ratio of $\text{Co}^{2+}/\text{Co}^{3+}$ and $\text{Ni}^{3+}/\text{Ni}^{2+}$ calculated from the fitted area are 1.20 and 2.22, respectively. It was reported that high ratio of Co²⁺ and Ni³⁺ contributed to the superior electrocatalytic activity for OER in the alkaline media [18,19]. Meanwhile, after the electrodeposition of MnOx NFs, the entire surface of NiCo₂O₄ NWs is covered by MnO_x NFs since the peaks are disappeared in the Co 2p spectrum of NiCo₂O₄@MnOOH/SSL. Simultaneously, the raw Mn 2p spectrum emerges in the NiCo₂O₄@MnO_y/SSL sample and can be fitted with two peaks: Mn^{3+} species (641.8 eV for Mn $2p_{3/2}$) and Mn⁴⁺ species (642.9 eV). There is also a satellite peak in the deconvolution spectrum of Mn 2p.

High electronic conductivity is also important for OER catalysts since the OER occurs on the electrode surface and the generated electrons thereafter need to be transferred. The electrical resistance of asprepared samples is measured in a self-designed cell. From the I-V curve as presented in Fig. 2d, it is observed that the deposition of nanostructured catalysts slightly increases the area resistances, which is still acceptable for practical application.

The electrochemical activity is measured by polarization curves, as shown in Fig. 3a. Due to the SSL mesh exhibits a poor activity toward OER process, the influence from the conductive substrate can be eliminated. In both NiCo2O4/SSL and NiCo2O4@MnOx/SSL, there exists a Co(III)/Co(IV) redox peak in prior to the OER process. While after being deposited with MnOx NFs, the onset potential for the OER is significantly negative-shifted, which indicates the MnO_x NFs contribute to the increased OER activity. Thus, to reach the same current density of 10 mA cm^{-2} , the overpotential for NiCo₂O₄@MnO_x/SSL is 342 mV, which is lower than that of overpotential for NiCo₂O₄/SSL (386 mV). Although the overpotential of NiCo₂O₄@MnO_x/SSL is still much larger than that of overpotential for IrO2/C (221 mV), our freestanding electrode outperforms other integrated electrodes [7,20] and NiCo₂O₄based hybrid catalysts [8,21] for the OER. The corresponding Tafel plots derived from LSV curves are plotted in Fig. 3b to gain further insight into the OER activity. Apparently, the overpotential for NiCo₂O₄@MnO_x/SSL is around 50 mV lower than that of NiCo₂O₄/SSL at the same current density. Simultaneously, the Tafel slope of NiCo₂O₄@MnO_x/SSL is 64 mV dec⁻¹, compared to 73 mV dec⁻¹ for NiCo₂O₄/SSL. Since the Tafel slope relates to the reaction mechanism, the reduced Tafel slope indicates the effective charge-transfer and high

Fig. 1. Typical SEM images of NiCo₂O₄/SSL (a) and NiCo₂O₄@MnO_x/SSL (b–c); a low-resolution TEM image of NiCo₂O₄@MnO_x/SSL (d); high-resolution TEM images of NiCo₂O₄ (e) and MnO_x (f).



efficiency of NiCo₂O₄@MnO_x/SSL for OER. The Tafel slope of NiCo₂O₄@MnO_x/SSL is similar to the value obtained on IrO₂/C (58 mV), which is indicative of the similar reaction mechanism.

Finally, the catalysts are employed to fabricate MEA and evaluated the electrochemical activity for AEM water splitting. As shown in Fig. 3c, with the aid of MnO_x NFs, the activation loss in the cell with NiCo₂O₄@MnO_x/SSL as the catalyst is significantly reduced as compared with the cell with NiCo2O4/SSL. At a current density of 200 mA cm^{-2} , the applied voltage of the cell decreases from 1.79 V to 1.74 V when the MnO_x NFs are deposited on the NiCo₂O₄ NWs. The improvement of electrocatalytic performance can be attributed to the synergistic effect. That is, the NiCo₂O₄ NWs contribute to the porous and electrically conductive skeleton, and the MnOx NFs contribute to the high electroactive sites as a result of the reduced activation barrier on the surface of MnO_x for the OER. Although the overpotential of the cell with NiCo₂O₄@MnO_x/SSL is still larger than that of the cell with IrO₂/C as the catalyst, the performance demonstrated here is competitive when compared with the performance reported in the literature [2,3]. Meanwhile, the electrode stability is determined by a continuous constant-current electrolysis (Fig. 3d). The cell voltage can be maintained at around 1.86 V at a current density of 400 mA cm⁻² for 100 h. The internal resistance was periodically measured by AC impedance at a high frequency (1 kHz). The internal resistance stays at around $0.42 \ \Omega \ cm^2$, which indicates that the as-prepared electrode is stable during the continuous operation.

4. Conclusions

In summary, NiCo₂O₄@MnO_x was supported on the SSL mesh for the AEM water electrolysis. It was demonstrated that the deposition of MnO_x NFs on the surface of NiCo₂O₄ NWs significantly enhanced the electrocatalytic activity toward the OER. More importantly, the AEM water splitting with the NiCo₂O₄@MnO_x/SSL as the OER catalyst was operated at 200 mA cm⁻² with an overpotential of about 500 mV and stably run for 100 h without significant degradation. These results suggest that the NiCo₂O₄@MnO_x can serve as a low-cost, high-performance catalyst for the OER in the alkaline environment.



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Fig. 2. The XRD patterns (a) of as-prepared samples. High-resolution XPS survey spectra for (b) Ni 2p; (c) Mn 2p; (d) area resistance measurement for different electrodes.

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Fig. 3. The *iR*-corrected LSV curves and Tafel plots for different electrodes (a and b); AEM water electrolysis performance of MEAs with different electrodes (c); transient voltage and AC resistance of AEM water splitting with NiCo₂O₄@ MnO_x/SSL as the electrode for OER (d).

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