

A Physical Model for Metal–Oxide Thin-Film Transistor Under Gate-Bias and Illumination Stress

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Abstract—A negative shift in the turn-on voltage of a metal–oxide thin-film transistor under negative gate-bias and illumination stress has been frequently reported. The stretched-exponential equation, predicated largely on a charge-trapping mechanism, has been commonly used to fit the time dependence of the shift. The fitting parameters, some with unsubstantiated physical origin, are extracted by curve fitting. A more physically based model is presently formulated, incorporating the photogeneration, transport, and trapping of holes. The model parameters of generation energy barrier, hole mobility, and trapping time constant are extracted from the measured gate-bias dependent turn-on voltage shift. It is theoretically deduced and experimentally verified that the degradation kinetics is either generation or transport limited. The model can be further applied to explain the attenuated shift under positive bias and illumination stress, if the screening of the electric field emanating from the gate bias is also accounted for. From the effects of asymmetric source/drain bias applied during stress, it is deduced that the trapping is localized along the length of the channel interface. The turn-on voltage of a transistor after such stress is constrained by the portion of the channel exhibiting the smallest shift.

Index Terms—Asymmetric stress, bias illumination stress, generation or transport limited, indium–gallium–zinc oxide (IGZO), reliability, thin-film transistor (TFT).

I. INTRODUCTION

WITH their relatively lower process temperature, higher field-effect mobility, lower leakage current, and higher transparency [1], metal–oxide–semiconductors such as zinc oxide and its variants are being pursued as promising

alternatives to amorphous silicon for the construction of thin-film transistors (TFTs) in the next-generation flat-panel displays. However, reliability issues of metal–oxide TFTs, particularly those related to gate-bias stress under illumination [2], and their underlying mechanism must be better understood and resolved before the technology is ready for wider industrial adaptation [3].

Among the variants of zinc oxide, indium–gallium–zinc oxide (IGZO) has been most intensely studied. TFTs based on IGZO have been reported to suffer the severest degradation under negative gate-bias and illumination stress (NBIS) [4], as revealed by the dependence of the shift (ΔV_{on}) of the turn-on voltage (V_{on}) on the stress duration (t) [5]. The negative ΔV_{on} during NBIS is usually attributed to the trapping of positive charges at or adjacent to the channel/insulator interface [6], [7].

Much has been reported on the analysis of the charge-trapping process by fitting the t -dependence of ΔV_{on} with a stretched-exponential equation [2], [8], [9] that is parameterized by an extrapolated “saturation” ΔV_{∞} at $t = \infty$, a characteristic trapping time constant τ , and a stretched-exponential factor β [10]

$$\Delta V_{on} = \Delta V_{\infty} [1 - e^{-(t/\tau)^{\beta}}]. \quad (1)$$

However, being a phenomenological fit rather than a physical model, the equation sheds little light on the rich character of the physical processes taking place during stress [4], [10], [11]. Examples of the diverse instability behavior include the different dependence of ΔV_{on} on illumination with different wavelengths [11], on the magnitude of the bias, and on positive bias and illumination stress (PBIS) [4]. It is likely such limitation arises from an emphasis only on the charge-trapping dynamics, but ignoring the kinetics of the photogeneration and transport of the responsible charge carriers.

In this paper, IGZO TFTs with thermally induced source/drain (S/D) regions [12] were fabricated to investigate device stability against bias and illumination stress. A simple 1-D “generation-transport-trap” model is proposed for NBIS-induced instability, accounting for the photogeneration and transport of holes across the thickness of the channel, and their eventual trapping at or near the channel/gate-insulator (GI) interface. Model parameters such as energy barrier for generation ($q\phi$), generation rate prefactor (g_0), hole mobility (μ_h), and interfacial trapping time constant (τ_s) are extracted from the measured dependence of ΔV_{on} on t

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and the gate bias (V_g) during stress. The effects of drain bias (V_d) on NBIS-induced instability were also studied. It is deduced that the hole trapping during stress is localized along the length of the channel interface. Consequently, the overall V_{on} of the transfer characteristics after stress is constrained by the portion of the channel suffering the least amount of trapping, thus exhibiting the smallest ΔV_{on} .

II. MODELING AND EXPERIMENTS

TFT fabrication started with the sputter deposition and patterning of ~ 80 -nm molybdenum (Mo) as the bottom gate electrode on an oxidized silicon wafer. A GI stack consisting of 50-nm silicon nitride topped with 75-nm silicon oxide (SiO_x) was deposited in a plasma-enhanced chemical vapor deposition (PECVD) reactor before ~ 20 -nm IGZO was deposited by sputtering from a target with a molar ratio of $\text{In}_2\text{O}_3:\text{Ga}_2\text{O}_3:\text{ZnO} = 1:1:1$. The active island was subsequently patterned and capped with 300-nm gas-permeable SiO_x passivation layer (PL) deposited in the same PECVD apparatus. After the S/D and gate contact holes were opened, gas-impermeable S/D metal electrodes consisting of a stack of sputtered aluminum (Al) on Mo were patterned to partially overlap the gate electrode. The TFT was then “activated” at 400°C for 4 h in an oxygen atmosphere to thermally induce the formation of the highly conductive S/D regions [13]. Shown in Fig. 1(a)–(c) are, respectively, a schematic of the resulting IGZO TFT, the cross section of the region of the TFT used in numerical simulation, and the coordinate system used to set up the model equations.

The TFTs were subjected to a variety of stress conditions, including pure illumination stress (IS) without bias, negative/positive bias stress (N/PBS) without and with green (~ 532 nm) or blue (~ 485 nm) illumination (green-NBIS or blue-N/PBIS). The intensity of the illumination was fixed at 0.4 W/m^2 , measured using a calibrated photodiode. For both N/PBS and N/PBIS, the V_g during stress was $-20/+20\text{ V}$, with the S/D electrodes grounded ($V_s = V_d = 0\text{ V}$). The transfer characteristics were measured in the dark after each illumination session using an Agilent 4156C Semiconductor Parameter Analyzer. The delay between the turning off of the light source and the electrical measurement was less than 1 s. V_{on} is defined as the V_g at which an exponential increase in the drain current (I_d) is first observed.

It is clear from Fig. 2 that blue-NBIS was the only stress configuration exhibiting a continuous negative shift in V_{on} with t . The absence of any significant deterioration of the pseudosubthreshold slope is a strong indication that no new defect states in the bandgap were created during the stress [14]. Because of the invariant transfer characteristics obtained during and after blue-IS and blue-PBIS, and in agreement with the reported fast decay of photocurrent in IGZO [15], long-living excess photogenerated electrons in the channel were eliminated as the cause of the negative ΔV_{on} [16] for blue-NBIS. Furthermore, the minimally affected V_{on} after PBIS and PBS reflects a good channel/GI interface relatively free of electron trap states.

The negative ΔV_{on} under blue-NBIS has properly been attributed to the trapping of photogenerated positive charges at

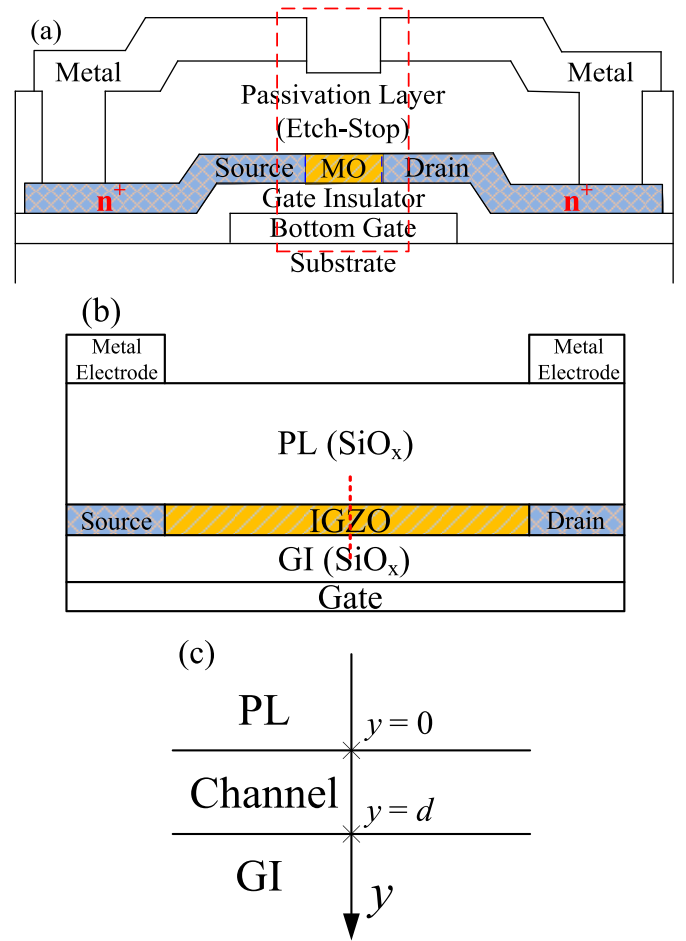


Fig. 1. (a) Schematic of IGZO TFTs with thermally induced and highly conductive S/D regions (cross hatched). (b) Cross section of the portion of a TFT [box bound by dotted line in (a)] under simulation; the electric field along the red line in the middle of the channel is simulated. (c) Coordinate system used to set up the model equations.

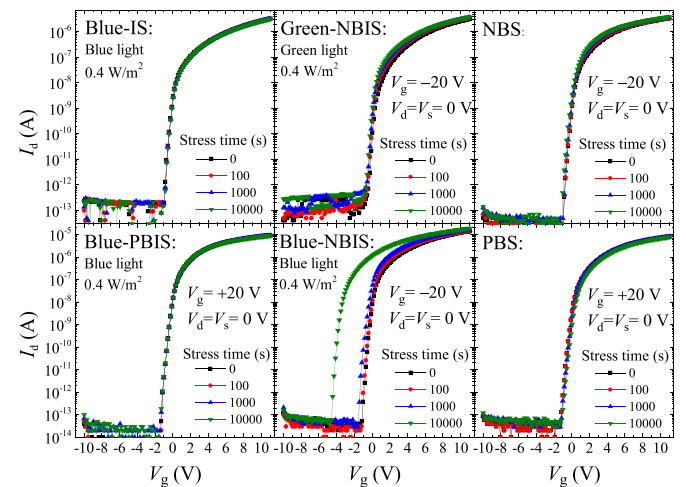


Fig. 2. Time evolution of the transfer characteristics of IGZO TFTs subjected to a variety of stress conditions including IS, N/PBS, and N/PBIS.

or near the interface between the IGZO channel and the SiO_x GI [6], [7], [17], [18]. Both ionized ($\text{V}_\text{O}^{2+/+}$) oxygen vacancy (V_O) defects and holes (h^+) have been proposed as possible

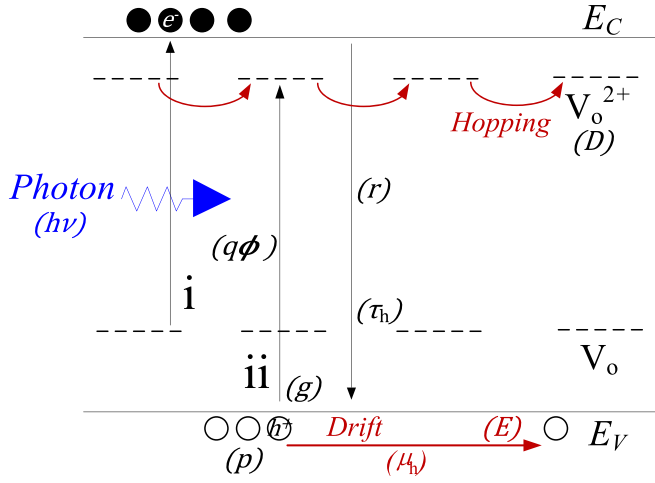
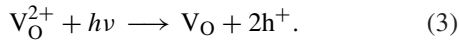
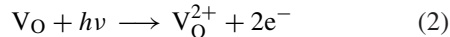


Fig. 3. Energy band diagram describing the photoassisted generation of ionized oxygen vacancies and holes. E_C and E_V are the respective edges of the conduction and valence bands.

candidates for such positive charges. Deduced by combining the Einstein relationship with the reported diffusion coefficient and activation energy [19], the room-temperature mobility of an ionized V_O is merely $\sim 10^{-27} \text{ cm}^2/\text{V} \cdot \text{s}$. This is ~ 22 orders of magnitude smaller than the theoretically predicted maximum μ_h of $\sim 10^{-5} \text{ cm}^2/\text{V} \cdot \text{s}$ [20]. Consequently, h^+ rather than V_O^{2+} are the more likely species accounting for the ΔV_{on} under blue-NBIS.

Holes are photogenerated via a two-step process, as summarized in (2) and (3), and shown schematically in Fig. 3.

- 1) Photoionization of V_O resulting in the promotion of electrons (e^-) to the conduction band and the formation of V_O^{2+} . This requires a photon energy of $\sim 2.3 \text{ eV}$ [11], [21].
- 2) h^+ generation by the promotion of e^- from the valence band to the V_O^{2+} state. This requires an energy of $\sim 2.8 \text{ eV}$ [22], [23]



A model is presently formulated for NBIS-induced ΔV_{on} , based on the following sequence of events: 1) the generation of h^+ by photoexcitation; 2) the transport of these h^+ across the channel; and 3) the final trapping of the h^+ at the interface. It is assumed that the population of photogenerated h^+ is small compared to the background of field-induced charge carriers so as not to materially change the electric field (E) established in the channel by the V_g .

The photogeneration rate g of h^+ is given by

$$g = g_0 e^{\frac{-q\phi + h\nu}{k_B T}} \quad (4)$$

where g_0 is a constant prefactor related to the density D of V_O^{2+} , q is elemental charge, ϕ is generation potential barrier, h is Planck's constant, ν is the frequency of illumination, k_B is Boltzmann's constant, and T is the absolute temperature. Note that under illumination, the effective barrier against generation is reduced by the photon energy ($h\nu$). Consequently, the

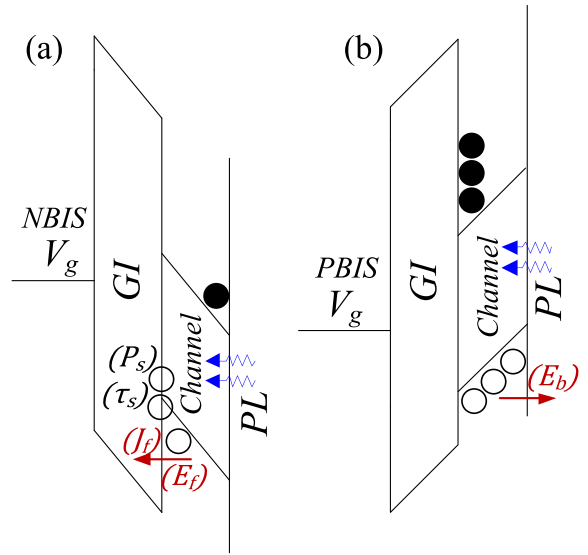


Fig. 4. Schematic of the degradation mechanism in TFTs under (a) NBIS and (b) PBIS.

Boltzmann factor alone contributes to a reduction of g by $\sim 10^4$ (originating from a reduction in $h\nu$ by $\sim 0.24 \text{ eV}$) at room temperature when switched from blue to green illumination. The recombination rate r of h^+ is expressed by

$$r = \frac{p}{\tau_h} \quad (5)$$

where p is the concentration of h^+ due to photogeneration and τ_h is the corresponding recombination time constant.

Given the much higher electron mobility ($\sim 10 \text{ cm}^2/\text{V} \cdot \text{s}$) than μ_h in IGZO [13], the photogenerated electrons can be transported rather more quickly than holes across the 20-nm channel thickness (d). Therefore, the continuity equation for only the holes needs to be considered during the stress

$$\frac{dp}{dt} = g - r - \frac{1}{q} \frac{dJ}{dy} \approx g_0 e^{\frac{-q\phi + h\nu}{k_B T}} - \frac{p}{\tau_h} - p\mu_h \frac{dE}{dy} = 0 \quad (6)$$

where $J = q\mu_h E p$ is the h^+ drift current density and y is defined in Fig. 1(c). Note that p is taken to be the total h^+ concentration due to the small intrinsic background concentration of holes in IGZO.

It is reported that $\tau_h \sim 10^{-2} \text{ s}$ [15], much shorter than the stress duration of 10000 s. Therefore, the local p is capable of quickly reaching steady state. For IS without bias, steady state $\Rightarrow g = r$ and $J = 0$. One deduces

$$p_0 = g_0 \tau_h e^{\frac{-q\phi + h\nu}{k_B T}} \quad (7)$$

where p_0 is the steady-state concentration of h^+ .

Under the action of E during NBIS, the photogenerated h^+ drift to and accumulate at the front interface between IGZO and GI. They are eventually trapped in the interface states. The total amount of trapped holes (P_s) is determined by a competition of the drift current density J_f being injected at the front interface and the detrapping process as shown in Fig. 4(a) such that

$$\frac{dP_s}{dt} = \frac{J_f}{q} - \frac{P_s}{\tau_s} \quad (8)$$

where τ_s is a time constant associated with the interfacial hole traps. It is similar to τ in (1) but different from τ_h in (5) for the recombination of the photogenerated electrons and holes in the bulk of the channel. Consequently

$$P_s = \frac{J_f}{q} \tau_s (1 - e^{-\frac{t}{\tau_s}}) = \mu_h E_f p_f \tau_s (1 - e^{-\frac{t}{\tau_s}}) \quad (9)$$

where E_f and p_f are, respectively, the electric field and the h^+ concentration at the front interface. A simple nonstretched exponential behavior is thus obtained

$$\Delta V_{on} = \frac{q P_s}{C_{ox}} = \frac{J_f}{C_{ox}} \tau_s (1 - e^{-\frac{t}{\tau_s}}) \quad (10)$$

where C_{ox} is the effective GI capacitance. Clearly, J_f is the factor determining ΔV_{on} . Two regimes of behavior regarding ΔV_{on} are discussed in the following.

When the electric field gradient dE/dy is large, such that r in (6) is negligible in comparison, the generated holes at each location across the channel are removed predominately by drift. Consequently, J_f is limited by generation. The g -limited J_{f1} is given by

$$J_{f1} = q d g_0 e^{\frac{-q\phi + h\nu}{k_B T}}. \quad (11)$$

Thus

$$\Delta V_{on1} = \frac{q d g_0}{C_{ox}} e^{\frac{-q\phi + h\nu}{k_B T}} \tau_s (1 - e^{-\frac{t}{\tau_s}}) \quad (12)$$

where ΔV_{on1} is the generation-limited ΔV_{on} and ΔV_{on1} is the corresponding saturation value.

For the opposite limiting case of dE/dy being negligible in comparison with r , p is dominated by the local equilibrium of generation and recombination, governed by (7). Consequently

$$J_{f2} = q \mu_h E_f p_0. \quad (13)$$

The transport-limited ΔV_{on2} is given by

$$\Delta V_{on2} = \frac{q \mu_h E_f g_0 \tau_h}{C_{ox}} e^{\frac{-q\phi + h\nu}{k_B T}} \tau_s (1 - e^{-\frac{t}{\tau_s}}) \quad (14)$$

where ΔV_{on2} is the corresponding saturation value. Which one of the two mechanisms dominates the t -dependence of ΔV_{on} under NBIS is determined by the relative magnitude of $1/\tau_h$ and $\mu_h dE/dy$ and ultimately the magnitude of V_g .

With the S/D electrodes grounded and the blue illumination power density fixed at 0.4 W/m^2 , the dependence of NBIS induced instability on the magnitude $|V_g|$ between 0 and 30 V was investigated. Two distinct regimes can be observed (Fig. 5) for the dependence of ΔV_{on} on V_g . When $|V_g|$ is lower than 8 V, the ΔV_{on} after $t = 10000 \text{ s}$ depends almost linearly on V_g . In this regime, E is relatively small, thus it is the limiting factor in regulating J_f . For $|V_g|$ greater than 8 V, a relatively constant ΔV_{on} was obtained, despite almost quadrupling $|V_g|$ from 8 to 30 V. This is the regime in which P_s is limited by the generation of holes.

ΔV_{on} extracted in the g -limited regime at $V_g = -30 \text{ V}$ is plotted in Fig. 6. For $t < 3000 \text{ s}$, the roughly linear

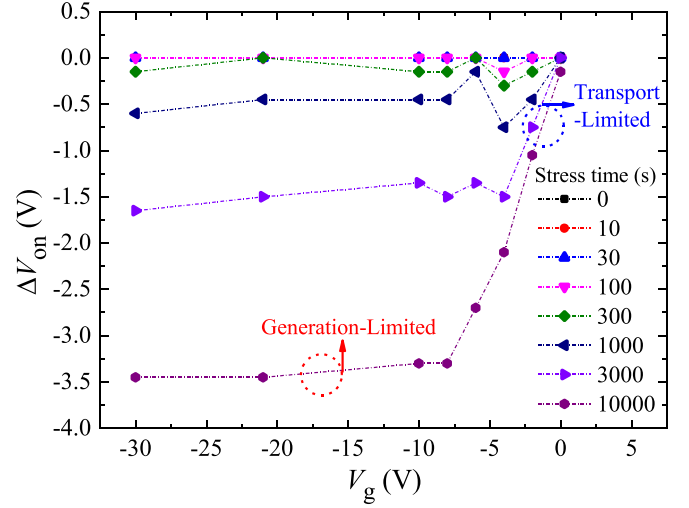


Fig. 5. Dependence of ΔV_{on} on V_g for IGZO TFTs subjected to NBIS for different t .

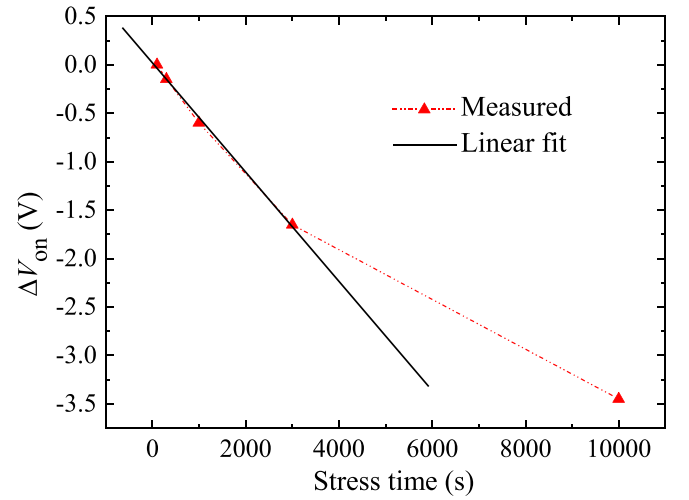


Fig. 6. Time evolution of ΔV_{on} during NBIS at $V_g = -30 \text{ V}$.

dependence of ΔV_{on} on t reflects the fact that the interface traps are far from being saturated with captured holes; hence, the corresponding saturation in Fig. 5 is a consequence of the limitation by h^+ generation in the bulk of the channel. For $t > 3000 \text{ s}$, the time rate of shift reduces as more of the trap states are filled, leading to a gradual saturation of ΔV_{on} when t continues to increase.

NBIS at elevated temperature (T) up to $T = 80^\circ \text{C}$ was investigated, with V_g fixed at -21 V and using the same illumination condition. This is also commonly known as the negative bias temperature and illumination stress (NBTIS). Shown in Fig. 7 is the t -dependence of the resulting ΔV_{on} . Included also are the analytical fits using the common stretched-exponential form according to (1) and the g -limited simple exponential form according to (12). Both fits are reasonable, with the corresponding average τ_s fluctuating around ~ 6000 and $\sim 7500 \text{ s}$ (inset of Fig. 7), thus indicating a relatively long-trapping time constant.

τ_s extracted at different T is shown in Fig. 8(a), lacking a clear dependence on T . The corresponding relative fluctuation

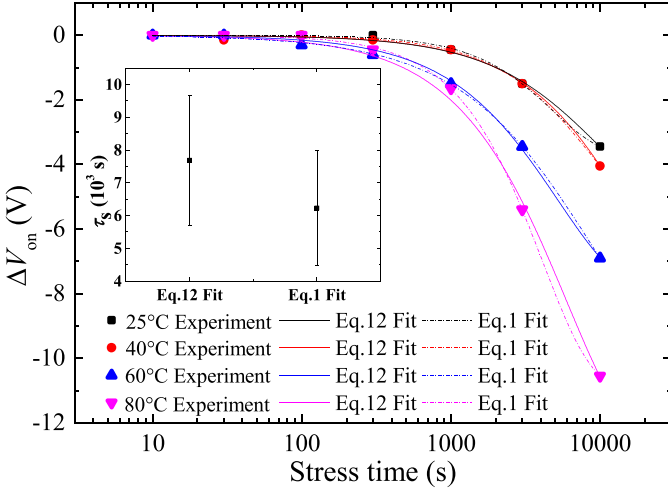


Fig. 7. t -dependence of ΔV_{on} after NBTIS at $V_g = -21$ V, and analytical fits using the generation–transport–trap model (12) and stretched-exponential equation (1). Inset: extracted τ_s for both fits.

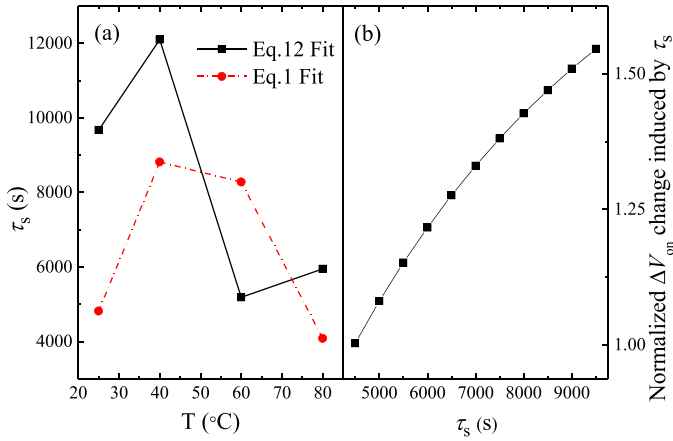


Fig. 8. (a) Variation of extracted τ_s with T . (b) Normalized ΔV_{on} change induced by the variation of extracted τ_s .

of $\tau_s(1 - e^{-(t/\tau_s)})$, thus its contribution to ΔV_{on} variation according to (12), is shown in Fig. 8(b). The variation of about $\sim 50\%$ for τ_s ranging from 5000 to 9000 s is small compared to the 400% increase of ΔV_{on} when T was increased from 25 °C to 80 °C. The larger variation should properly be assigned to the $e^{(-q\phi + h\nu/k_B T)}$ exponential term in $\Delta V_{\infty 1}$. Consequently, the average τ_s of 7500 s is used for the extraction of the model parameters.

Shown in Fig. 9(a) is the T -dependence of ΔV_{on} after 10000-s NBTIS in linear scale and in Fig. 9(b) is the same in Arrhenius plot. Higher temperature obviously enhances the generation rate of holes [24]. From the slope of the plot in Fig. 9(b), an energy barrier $q\phi$ of 2.75 eV is obtained. This is close to the reported energy difference of ~ 2.8 eV [22], [23] between the valence band edge and the V_O^{2+} level. Because of the large $q\phi$ compared with the thermal energy (~ 31 meV) even at 80 °C, it is small wonder that ΔV_{on} is negligible under NBTIS, as shown at the top of Fig. 8(a).

Since $V_{on} \sim 0$ V, E_f can be approximated by assuming the entire V_g is dropped across the thickness (d_{ox}) of the GI due

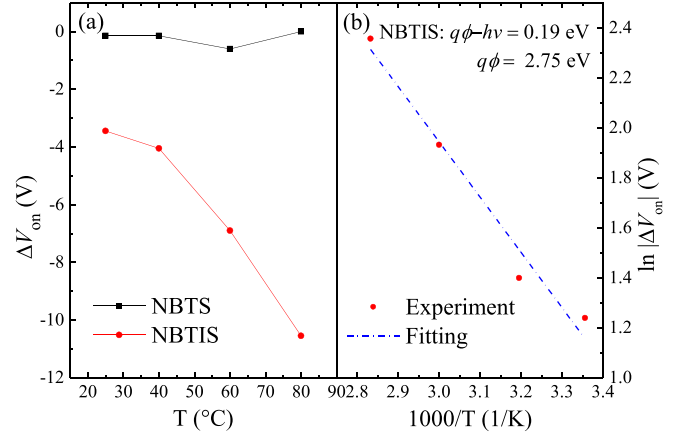


Fig. 9. (a) Dependence of ΔV_{on} on T for IGZO TFTs subjected to NBTIS and NBTIS. (b) $\ln |\Delta V_{on}|$ as a function of $1000/T$.

TABLE I

EXTRACTED PARAMETERS OF GENERATION–TRANSPORT–TRAP MODEL BASED ON MEASURED RELIABILITY PERFORMANCE OF FABRICATED IGZO TFTs

Parameters	Unit	Value
Generation energy barrier $q\phi$	eV	~ 2.75
Trap time constant τ_s	s	~ 7500
Holes mobility μ_h	cm^2/Vs	$\sim 10^{-9}$
Generation rate pre-factor g_0	$/\text{cm}^3\text{s}$	$\sim 10^{17}$
Generation rate g	$/\text{cm}^3\text{s}$	$\sim 10^{14}$
Steady-state hole concentration p_0	$/\text{cm}^3$	$\sim 10^{12}$

to interfacial band pinning in IGZO, i.e

$$E_f \approx \frac{V_g}{d_{ox}} \frac{\epsilon_{ox}}{\epsilon_{IGZO}} \quad (15)$$

where ϵ_{ox} and ϵ_{IGZO} are the respective effective dielectric constants of the GI and IGZO. The same extraction procedure is next applied to the set of ΔV_{on} obtained in the transport-limited regime shown in Fig. 5, this time using (14). The extracted model parameters are summarized in Table I.

The different behavior of N/PBIS was studied by estimating the distribution of E across d (Fig. 10) using a commercial device simulator and the schematic device cross section shown in Fig. 1(b). The highly conductive S/D regions, with a low resistivity of $\sim 10^{-2} \Omega \cdot \text{cm}$ [4], were treated as conductors. Other physical parameters and dimensions were unchanged.

Because of the change in the direction of E , photogenerated holes are transported to the IGZO/PL back interface during PBIS. It is clear that $|E_b|$, the magnitude of the electric field at the back interface during PBIS is significantly attenuated (by ~ 500 times at $|V_g| = 20$ V) compared to $|E_f|$,

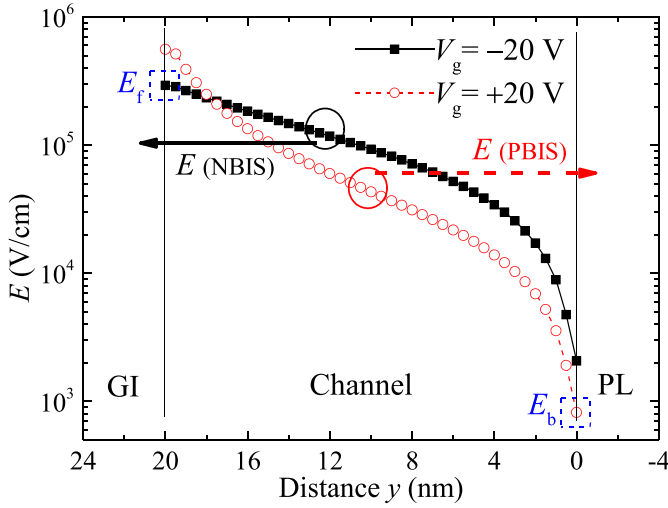


Fig. 10. Simulated distribution of E across the channel at $V_g = \pm 20$ V. E_f/E_b is responsible for the transport of holes near the corresponding interfaces under N/PBIS, respectively.

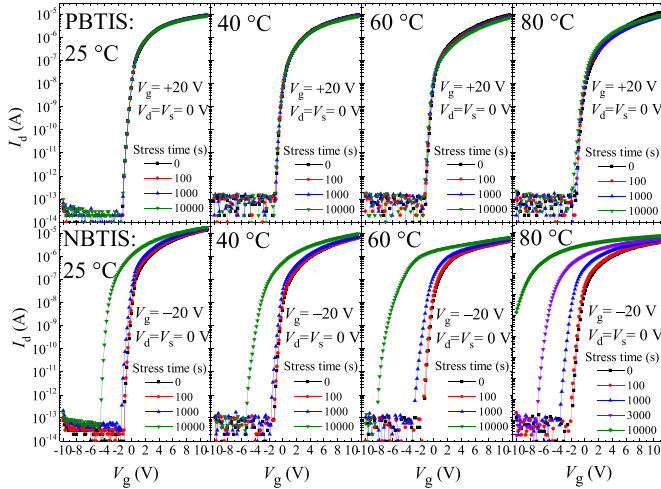


Fig. 11. Comparison of the time evolution of the transfer curves of TFTs subjected to P/NBTIS.

the magnitude of the electric field at the front interface during NBIS. The weak E_b leads to a slow migration, hence collection, of holes, thus a significant attenuation in the PBIS (Fig. 2)- and PBTIS-induced (Fig. 11) ΔV_{on} over t . The different band-bending configurations during N/PBIS have been schematically shown in Fig. 4. The simulated $|E_f| \sim 3 \times 10^5$ V/cm is close to the theoretical approximate of 5.8×10^5 V/cm given in (15).

III. EFFECTS OF ASYMMETRIC S/D BIAS

The forward and reverse families of transfer characteristics upon reversal of the S/D bias of an IGZO TFT subjected to NBIS with symmetrically grounded S/D regions are shown in Fig. 12. As expected, the two families are similar, with ΔV_{on} showing indistinguishable t -dependence.

It is clear from the schematic device cross section shown in Fig. 1(a) that E is generally nonuniform across the length of the channel between the S/D regions. In fact, $|E|$ would be the smallest in the middle of the channel if both the S/D electrodes

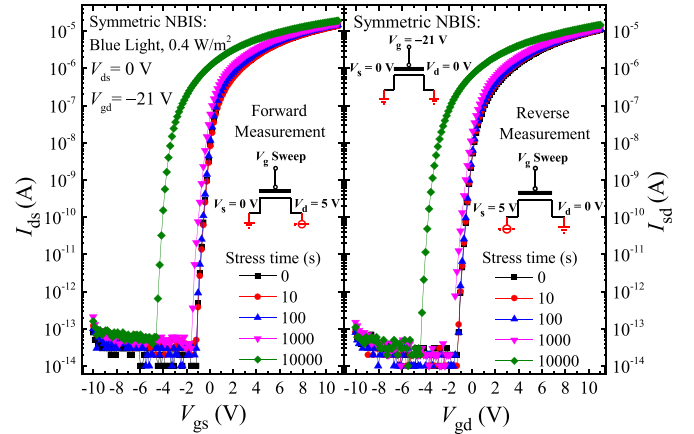


Fig. 12. Time evolution of the forward (Left: I_{ds} versus V_{gs} at $V_{ds} = 5$ V) and reverse (Right: I_{sd} versus V_{gd} at $V_{sd} = 5$ V) transfer curves subjected to NBIS with symmetric S/D bias.

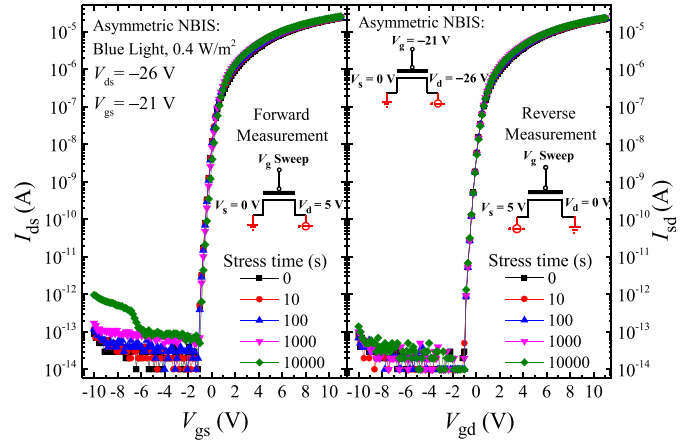


Fig. 13. Time evolution of the forward (Left: I_{ds} versus V_{gs} at $V_{ds} = 5$ V) and reverse (Right: I_{sd} versus V_{gd} at $V_{sd} = 5$ V) transfer curves subjected to NBIS with asymmetric S/D bias.

were grounded during NBIS. It can be deduced from (14) that ΔV_{on} reduces with decreasing $|E|$. Since $\Delta V_{on} < 0$ V, the turning on of a stressed TFT is thus controlled by the portion along the channel with the smallest $|\Delta V_{on}|$. In the case of a TFT subjected to symmetric NBIS with grounded S/D electrodes, the point of minimum stress is located in the middle of the channel.

The point of the minimum stress can be displaced from the middle of the channel by breaking the symmetry of grounded S/D during stress. With a grounded source $V_s = 0$ V but a $V_d = -26$ V during NBIS, the resulting forward and reverse families of characteristics are shown in Fig. 13. The two are still more or less identical, except $\Delta V_{on} \approx 0$ V. Since V_g was more positive than V_d during such stress, the drain end was actually subjected to PBIS. Consequently, it is this end that controls the turning on of the stressed TFT, hence the small $|\Delta V_{on}|$ of ~ 0 V. This behavior can be nicely captured using a circuit of serially connected TFTs shown in the inset of Fig. 14, with the limiting TFT near the drain end exhibiting a smaller ΔV_{on} .

Clearly, NBIS can be performed by setting V_d [5], [25] over a range that changes the stress condition at the drain end

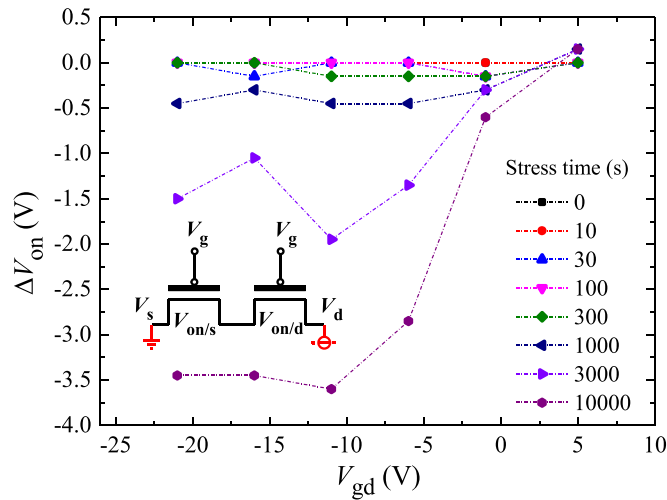


Fig. 14. Dependence of ΔV_{on} of TFTs under NBIS on the potential difference V_{gd} from gate to drain. Inset: circuit schematic of serially connected TFTs near the source and drain ends of the channel.

continuously from NBIS to PBIS. Such a detailed study has indeed been carried out, and the results are shown in Fig. 14, with V_g and V_s fixed, respectively, at -21 and 0 V, but V_d changed from 0 to -26 V.

It can be seen that for a given t , $|\Delta V_{on}|$ increases continuously as $V_{gd} \equiv V_g - V_d$ decreases from 5 to -35 V and saturates beyond $V_{gd} \approx -10$ V. This is consistent with the two regimes of degradation kinetics reported in Fig. 5.

IV. CONCLUSION

The stretched-exponential equation, predicated largely on a charge-trapping mechanism, has been commonly used to fit the time dependence of the shift in the turn-on voltage of a metal–oxide TFT under NBIS. Constrained by its emphasis on only the charge-trapping dynamics, the model cannot be used to fully account for the rich character of the physical processes taking place during the stress. A more physically based model is presently formulated, incorporating the photogeneration, transport, and trapping of holes. For indium–gallium–zinc oxide, the respective model parameters of generation energy barrier ~ 2.75 eV, hole mobility $\sim 10^{-9}$ cm²/V · s, and trapping time constant ~ 7500 s are extracted from the measured gate-bias dependent turn-on voltage shift. It is theoretically deduced and experimentally verified that the degradation kinetics is either generation or transport limited, depending on the magnitude and direction of the local electric field inside the channel but normal to the channel/interface during the stress. The model can be further applied to explain the attenuated shift under PBIS, when the screening of the electric field emanating from the gate bias has been accounted for. From the effects of asymmetric S/D bias applied during a bias illumination stress, it is deduced that hole trapping is localized along the length of the channel interface. The turn-on voltage of a transistor after such stress is constrained by the portion of the channel exhibiting the smallest shift. Since the model is developed to describe an intrinsic degradation mechanism within the IGZO active

layer, it is believed that the proposed generation-transport-trap model is equally suitable for describing similar degradation in other structures built around similar active layers—if the structure induced variation in the distribution of electric field is accounted for.

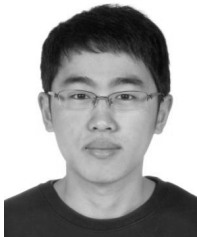
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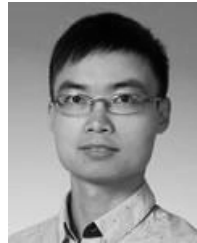
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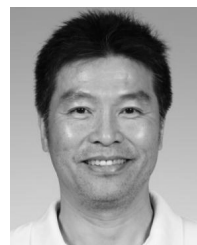
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